

Reviews

Inorganic Materials as Catalysts for Photochemical Splitting of Water

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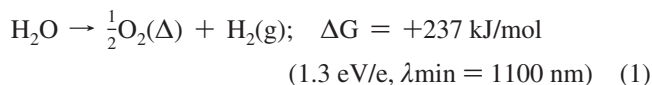
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Photochemical splitting of water into H₂ and O₂ using solar energy is a process of great economic and environmental interest. Since the discovery of the first water splitting system based on TiO₂ and Pt in 1972 by Fujishima and Honda, over 130 inorganic materials have been discovered as catalysts for this reaction. This review discusses the known inorganic catalysts with a focus on structure–activity relationships.

Introduction

At a power level of 1000 W/m², the solar energy incident on the earth's surface by far exceeds all human energy needs.^{1,2} Photovoltaic³ and electrochemical solar cells^{3–5} that convert solar energy into electricity can reach up to 55–77% efficiency^{6–8} but remain uneconomical because of high fabrication costs, insufficient light absorption,⁹ and inefficient charge transfer.³ In a process that mimics photosynthesis, solar energy can also be used to convert water into H₂ and O₂, the fuels of a H₂-based energy economy.



Reaction 1 is catalyzed by many inorganic semiconductors, the first of which, TiO₂, was discovered in 1971 by Fujishima and Honda (see below).^{10,11} Today, over 130 materials and derivatives are known to either catalyze the overall splitting of water according to eq 1 or cause water oxidation or reduction in the presence of external redox agents. Current record holders in terms of quantum efficiencies (QEs) are NiO-modified La/KTaO₃ (QE = 56%, pure water, UV light),¹² ZnS (QE = 90%, aqueous Na₂S/Na₂SO₃, light with $\lambda > 300 \text{ nm}$),¹³ and Cr/Rh-modified GaN/ZnO (QE = 2.5%, pure water, visible light).^{14,15} So far, no material capable of catalyzing reaction 1 with visible light and a QE larger than 10% has been found. Here, 10% is the limit for commercial applications.¹⁶ In order to highlight recent developments and to identify promising directions in this increasingly complex research area, this review summarizes the known inorganic catalysts and discusses property–activity relationships with a focus on structural features. After a brief introduction of the basic physical concepts, catalysts are presented starting with the transition element oxides, followed by the main-group element oxides, nitrides, and phosphides, and the sulfides. The review concludes with a brief discussion of

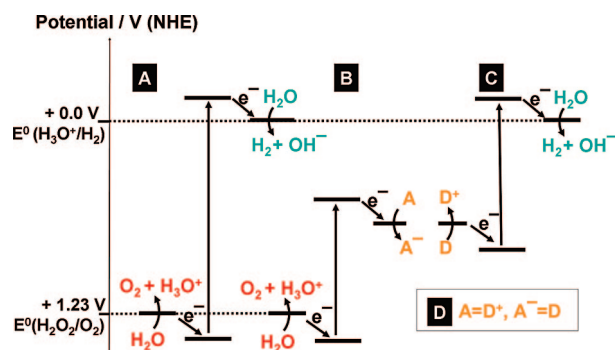


Figure 1. Potential energy diagrams for photochemical water splitting at pH = 0: (A) single semiconductor system; (B) with an electron acceptor; (C) with an electron donor; (D) dual semiconductor system (z scheme) employing a redox shuttle. Modified from ref 108.

important criteria for effective catalysts. Readers interested in other aspects of solar-energy utilization and photocatalysis are referred to a series of excellent review articles on renewable energy,^{1,17,18} general photocatalysis,^{19–21} water oxidation catalysts,^{22,23} photosystem II,²⁴ molecular,^{25,26} colloidal,^{27–32} and heterogeneous water splitting catalysts,^{33–46} including transition-metal oxides,^{38–43,47–49} metal oxynitrides,^{44–46} and photoelectrochemical cells and their physical principles.^{8,16,18,50–61}

General Principles

The photocatalytic properties of inorganic semiconductors strongly depend on the electronic band structure. For photochemical water reduction to occur, the flat-band potential of the semiconductor (for highly doped semiconductors, this equals the bottom of the conduction band) must exceed the proton reduction potential of 0.0 V vs NHE at pH = 0 (−0.41 V at pH = 7; Figure 1A). To facilitate water oxidation, the potential of the valence band edge must exceed the oxidation potential of water of +1.23 V vs NHE at pH

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= 0 (+0.82 V at pH = 7). Based on these parameters, a theoretical semiconductor band-gap energy of ~ 1.23 eV is required to drive the water-splitting reaction according to eq 1. The smallest band gap achieved so far in a functional catalyst is 2.30 eV in $\text{NiO/RuO}_2\text{-Ni:InTaO}_4$.^{47,62} Semiconductors with smaller band gaps or lower flat-band potentials require a bias voltage or external redox reagents to drive the reaction (Figure 1B/C). Alternatively, two or more small-band-gap semiconductors can be combined to drive water oxidation/reduction processes separately via multiphoton processes (Figure 1D).

Surprisingly, literature values for flat-band potentials^{35,50,55,63–68} exhibit large variations; e.g., reported values for Ta_2O_5 range from -0.3 ⁶⁹ to -1.4 V.⁶⁴ Discrepancies reflect experimental uncertainties⁶⁹ and intrinsic differences resulting from variable material preparations. It is well-known that flat-band potentials strongly depend on ion absorption (protonation of surface hydroxyl groups), crystallographic orientation of the exposed surface, surface defects, and surface oxidation processes (sulfur on CdS).^{70,71} These and other factors^{55,72} are rarely considered in the preparation and testing of photochemical water-splitting catalysts. For the design of new materials, it can therefore be useful to estimate the flat-band potentials with a method developed by Butler and Ginley.⁷³ If the geometric mean χ of the Mulliken electronegativities of the semiconductor constituents is known, the flat-band potential V_{fb} [V vs NHE] for a metal oxide can be calculated as

$$V_{\text{fb}} = E_0 - \chi + \frac{1}{2}E_{\text{G}} \quad (2)$$

Here E_{G} is the semiconductor band gap [eV] and E_0 (+4.44 eV) is the energy of a free electron on the H_2 redox scale.^{60,73,74} Activities of photochemical water-splitting catalysts are usually assessed with the rates of evolved gases [mol/h] per catalyst amount [g] under the specified irradiation conditions. From the measured evolution rate [H_2], the apparent QE = $2[\text{H}_2]/I$ of the catalyst can be calculated using the known photon flux I [mol/s] incident on the reaction mixture (as determined by, e.g., ferrioxalate actinometry⁷⁵). If available, this information is included with the experimental conditions in Table 1. The structures of selected semiconductors are shown in Figures 2–4.

TiO₂

The photocatalytic properties of TiO_2 have been reviewed previously.^{16,27,29,32,33,76–78} Titania (TiO_2) was the first material described as a photochemical water-splitting catalyst. It crystallizes in three structure types: Rutile, Anatase, and Brookite. All modifications contain TiO_6 octahedra that are interconnected via two (Rutile), three (Brookite), or four (Anatase) common edges and via shared corners, and as a result, the band gaps (3.0 eV for Rutile and 3.15 eV for Anatase) differ slightly.⁶⁷ In their 1971/72 papers, Fujishima and Honda described an electrochemical cell consisting of a n-type TiO_2 (Rutile) anode and a Pt black cathode.^{10,11} When the cell was irradiated with UV light (<415 nm) from a 500 W Xe lamp, O_2 evolution takes place at the anode and a current flows to the Pt counter electrode. Based on the current, a photoelectrochemical efficiency of $\sim 10\%$ was

estimated. While formation of H_2 at the cathode was not confirmed in the original paper, Wrighton et al. showed in 1975 that O_2 and H_2 were indeed formed under similar conditions and that they were both from water.⁷⁹ In the same year, the first report of photocatalytic water splitting by a powdered TiO_2 catalyst appeared.⁸⁰ When wetted TiO_2 powder (Table 1) was exposed to water vapor under UV/vis irradiation, H_2 and O_2 were evolved at a near stoichiometric ratio. In the presence of N_2 , NH_3 was produced, while HCCH gave CH_4 , C_2H_4 , and C_2H_6 . Thermal annealing and storage conditions greatly affected the activity of the catalyst, which was most active as a mixture of 23% Rutile and 77% Anatase. These findings were disputed by Van Damme and Hall,⁸¹ who attributed the observed reactivity to photolytic decomposition of hydroxyl groups on the TiO_2 surface, as opposed to a catalytic process. However, Kawai and Sakata demonstrated only 1 year later that catalytic amounts of D_2 were produced when a heat-treated TiO_2 powder was irradiated in the presence of D_2O vapor.⁸² No O_2 was evolved when pure Rutile was used, but the addition of RuO_2 led to a functional catalyst that produced both D_2 and O_2 after the initial formation of CO as a side product.

In the late 1970s, titania catalysts were improved significantly following ideas that were first expressed by Nozik.^{55,83} In his 1977 paper, Nozik proposed the concept of “photochemical diodes” (later termed antenna catalysts⁸⁴), which consisted of two electrodes from a photoelectrochemical cell directly fused together. This produced either p–n-type (e.g., p-GaP/n- TiO_2) or Schottky-type devices (e.g., n- CdS/Pt or p-GaP/Pt). The efficiency of such dual-component catalysts was expected to improve because a space charge layer at the material interface enhances electron–hole separation. Accordingly, in 1980, Sato and White found that enhanced evolution of H_2/O_2 took place on a TiO_2 catalyst to which a Pt cocatalyst had been added via photodecomposition of hexachloroplatinate in an acetic acid solution.⁸⁵ In this study, ^{18}O -labeling experiments also confirmed O_2 production from water. Without Pt, no H_2 was formed on Anatase TiO_2 . However, as a drawback, the Pt portion in the material catalyzed the reverse of the water-splitting reaction according to eq 1, limiting yields at high H_2 partial pressure. Following earlier work on molecular dyes,^{86–88} Graetzel’s group published a series of papers in the 1980s on multicomponent catalysts comprised of colloidal TiO_2 (Anatase) particles doped with 0.4% of Nb_2O_5 and joined together with Pt metal and with RuO_2 particles.^{29,89–92} Under UV irradiation, aqueous suspensions of these particles produced stoichiometric amounts of H_2 and O_2 . Here, TiO_2 generates the electron–hole pair, of which the electron is subsequently transferred to the Pt water reduction site and the hole to the RuO_2 water oxidation site.⁹² Again, slow recombination of H_2 and O_2 took place in the dark because of the Pt cocatalyst. By the addition of $[\text{Ru}(\text{bpy})_3]^{2+}$ as a sensitizer,⁹² the system was able to split water under visible light irradiation. Here, the electron–hole pair is generated on the $[\text{Ru}(\text{bipy})_3]^{2+}$ complex, and the role of the TiO_2 particle is merely to accept the electron from the sensitizer and to funnel it to Pt, where H_2 evolution occurs. The sensitizer is subsequently reduced at the RuO_2 site, which oxidizes water to O_2 . The overall

Table 1. Catalyst Overview

no.	material	cocatalyst	irradiation conditions	reaction details and refs
1a	TiO ₂		UV, 360 W Hg	H ₂ /O ₂ at 1.16/0.55 $\mu\text{mol/h}$ /0.2 g of cat. ⁸⁰
1b	TiO ₂	Pt	UV, 200 W Hg	H ₂ and O ₂ in stoich. ratio (0.1 $\mu\text{mol/h}$ of H ₂ per 250 mg of cat.. ⁸⁵
1c	Nb ₂ O ₅ :TiO ₂ (Anatase)	Pt and RuO ₂	450 W Xe at >400 nm	up to 5 mL/h/0.1 g of H ₂ (222 $\mu\text{mol/h}$ /0.1 g) from aqueous EDTA and up to 4.5 mL/h/0.05 g of H ₂ (200 $\mu\text{mol/h}$ /0.1 g) from water with stoich. O ₂ ; ⁸⁹ See also refs 29, 90, and 92; for Pt/RuO ₂ /TiO ₂ , QE up to 30% under UV ⁹¹ and up to 5% with [Ru(bpy) ₃] ²⁺ sensitizer under vis light ⁹²
1d	TiO ₂	Pt	200 W Hg at 440 nm	3–4 mL of H ₂ after 5 h; pH = 4.78 using Ru(bipy) ₃] ²⁺ , MV ²⁺ , as an electron relay and EDTA as a sacrif. donor; cat. < 1 mg ⁸⁴
1e	TiO ₂	Pd and Rh	UV, 500 W Xe	449 $\mu\text{mol/h}$ /0.3 g (QE = 29%) for Rh/TiO ₂ compared to 284 $\mu\text{mol/h}$ /0.3 g (QE = 17%) for Pt/TiO ₂ ⁹⁵
1f	TiO ₂ Anatase and Rutile	Pt	UV at >300 nm, 400 W Hg	H ₂ (180 $\mu\text{mol/h}$ /0.5 g) and O ₂ (90 $\mu\text{mol/h}$ /0.5 g) from water at pH = 11 (QE = 4%) ^{107,108}
1g	N-doped TiO ₂	Pt	450 W Xe at >420 nm	21 $\mu\text{mol/h/g}$ (QE = 14%) of O ₂ , traces of H ₂ ; AgNO ₃ as the acceptor ¹¹³
2a	SrTiO ₃	NiO	450 W Hg	stoich. water vapor splitting at 4.4×10^{-3} mL/h/2.0 g (H ₂); ¹¹⁶ different study, 100 μmol of H ₂ /h/g with stoich. O ₂ ¹¹⁷
2b	Cr/Ta:SrTiO ₃	Pt	300 W Xe at 420.7 nm	splits water into H ₂ /O ₂ at 0.21/0.11 $\mu\text{mol/h}$ in tandem with Pt–WO ₃ (0.4 g of both cat., QE = 0.1%); ¹¹⁸ Cr/Ta:SrTiO ₃ –Pt produced H ₂ at 0.8 $\mu\text{mol/h}$ /0.2 g from KI(aq); Pt–WO ₃ alone with NaIO ₃ produced O ₂ at 84 $\mu\text{mol/h}$ /0.2 g
2c	SrTiO ₃	Ru, Ir, Pd, Pt, Os, Re, Co	1000 W Xe/Hg at >300 nm	stoich. amounts of H ₂ /O ₂ from pure water; all values for 0.05 g of cat. are Rh/SrTiO ₃ = 628 μmol L/h, Ru/SrTiO ₃ = 159 μ L/h, Re = 107 μmol L/h, Ir = 80 μ L/h, Pt = 107 μmol L/h, Pd = 71 μ L/h, Os = 62 μmol L/h, Co = 26 μ L/h, SrTiO ₃ = 11 μmol L/h of H ₂ but no O ₂ ^{119,120}
3, 4	La ₂ TiO ₅ , La ₂ Ti ₃ O ₉	NiO	UV at 450 W Hg	H ₂ /O ₂ in stoich. ratio at 386 $\mu\text{mol/h/g}$ (La ₂ Ti ₃ O ₉) and 442 $\mu\text{mol/h/g}$ (La ₂ TiO ₅) ¹²¹
5a	La ₂ Ti ₂ O ₇	NiO	UV at 450 W Hg	307 and 152 $\mu\text{mol/h}$ /0.5 g of H ₂ /O ₂ ; ¹²³ another study reports 441 $\mu\text{mol/h/g}$ of H ₂ with stoich. O ₂ (QE = 12%); ¹²² doping with BaO and addition of NaOH gives 5 mmol of H ₂ /h/g with stoich. O ₂ ; ¹²¹ H ₂ at rates of 120 (Pt cocat.) and 400 $\mu\text{mol/h/g}$ (NiO, O ₂ not measured); ¹²⁴ hydrothermal synthesis: 72.5 μmol of H ₂ /h/g (500 W Hg, O ₂ not determined) ¹²⁵
5b	La ₂ Ti ₂ O ₇ , doped with Fe, Cr	Pt	500 W Hg at >420nm	H ₂ at <15 $\mu\text{mol/h/g}$ from MeOH(aq) ¹²⁴
6	Sr ₃ Ti ₂ O ₇		UV, 400 W Hg	~20 $\mu\text{mol/h/g}$ of H ₂ ; with NiO; 77 $\mu\text{mol/h/g}$ with stoich. O ₂ ; rate increase for NiO-modified material from ester pyrolysis 144 $\mu\text{mol/h/g}$ with stoich. O ₂ evol. ¹²⁶
7	PbTiO ₃	Pt	450 W Xe at >420 nm	13.6/523 $\mu\text{mol/h}$ /0.3 g of H ₂ /O ₂ from aqueous MeOH and AgNO ₃ ¹²⁷
8	Sm ₂ Ti ₂ S ₂ O ₅	Pt	300 W Xe at 440–650 nm	38 $\mu\text{mol/h}$ /0.2 g (QE = 0.1%) of H ₂ with Na ₂ S–Na ₂ SO ₃ or methanol; Sm ₂ Ti ₂ S ₂ O ₅ –IrO ₂ gives O ₂ at 11 $\mu\text{mol/h}$ /0.2 g (QE = 1.1%) from aqueous AgNO ₃ ¹²⁸
9–11	M ₂ La ₂ Ti ₃ O ₁₀ (M = K, Rb, Cs), doped with Nb	Pt	UV, 450 W Hg	stoich. H ₂ /O ₂ of 79 $\mu\text{mol/h/g}$ (for Rb ₂ La ₂ Ti ₂ NbO ₁₀) to 869 $\mu\text{mol/h/g}$ (for Rb ₂ La ₂ Ti ₃ O ₁₀ , QE = 5%); ^{129–131} activity of K ₂ La ₂ Ti ₃ O ₁₀ depends on the cocat. (Ni, Pt, or RuO ₂) and on the KOH conc. ¹³³ different synthesis gives 2.89 mmol of H ₂ /h/g and 1.13 mmol of O ₂ /h/g; ¹³⁴ Au-modified K ₂ La ₂ Ti ₃ O ₁₀ gave H ₂ at 841 $\mu\text{mol/h/g}$ from a 0.1 M KOH solution; O ₂ not meas. ¹³²
12	PbBi ₄ Ti ₄ O ₁₅	Pt	450 W Xe at >420 nm	11.2/433 $\mu\text{mol/h}$ /0.3g of H ₂ /O ₂ employing MeOH and AgNO ₃ as sacrif. agents. ¹²⁷
13	BaTi ₄ O ₉	RuO ₂	UV, 400 W Xe	H ₂ at 180 $\mu\text{mol/h/g}$ with stoich. O ₂ ^{42,259}
14–16	M ₂ Ti ₆ O ₁₃ (M = Na, K, Rb)	RuO ₂	UV, 400 W Xe	H ₂ /O ₂ rates highest for M = Na (17/8 $\mu\text{mol/h/g}$) ^{42,135}
17	La ₄ CaTi ₅ O ₁₇	NiO	450 W Hg at >320 nm	499 $\mu\text{mol/h/g}$ of H ₂ and stoich. O ₂ (QE = 20%) ¹²²
18	M ₃ (PO ₄) ₄ (M = Ti, Zr)	Pt	300 W Xe at >290 nm	Pt/Ti ₃ (PO ₄) ₄ (0.91 $\mu\text{mol/h/g}$), Pt/M ₃ (PO ₄) ₄ (M = 50% Ti, 50% Zr, 5.91 $\mu\text{mol/h/g}$) ¹³⁶
19	Ta ₂ O ₅	NiO _x , Pt, or RuO ₂	UV, 400 W Hg	190 $\mu\text{mol/h/g}$ with stoich. O ₂ yield ⁴⁰ and 1150/530 $\mu\text{mol/h/g}$ under optimized conditions ⁴⁸
20, 21	A ₄ Ta _x Nb _{6-x} O ₁₇ (A = Rb, K, x = 1–4)	NiO	400 W Hg	100–936 μmol of H ₂ /h/g of cat. (stoich. O ₂ evol.); activity decreases with increasing Ta content ¹⁴⁰
22–24a	MTaO ₃ (M = Li, Na, K)	NiO	UV, 400 W Hg	H ₂ and O ₂ from pure water at 3.39 and 1.58 mmol/h/g (QE = 20–28%, at 270 nm); ^{48,141,142} 10–40 times lower activity without cocat.; ¹⁴² sol–gel method produces active cat. with H ₂ /O ₂ 1940/~1000 $\mu\text{mol/h/g}$ ¹⁴³

Table 1. Continued

no.	material	cocatalyst	irradiation conditions	reaction details and refs
24b	La:NaTaO ₃	NiO	UV, 400 W Hg	H ₂ and O ₂ at 19.8 and 9.7 mmol/h/g, from pure water ¹² (QE = 56%)
25	La _{1/3} TaO ₃	NiO	UV, 400 W Hg	35/7.9 μmol/h/0.5 g of H ₂ /O ₂ from water ¹⁴⁴
24c	KTaO ₃ doped with Ti, Zr, Hf	NiO	500 W Xe	H ₂ /O ₂ from pure water at rates of up to 100/30 μmol/h/0.1 g; ¹⁴⁵ Ti works best (at 8% doping level), followed by Zr and Hf
24d	Zr:NaTaO ₃	Pt	UV, 500 W Xe	Zn, Co, Cr, and Mg porphyrin as the sensitizer; up to 57/28 μmol/h/0.1 g of H ₂ /O ₂ from pure water (for cyanocobalamin); QE up to 12.8% ¹⁴⁶
26a/b	Ni:InTaO ₄ , InTaO ₄	RuO ₂ or NiO	400 W Hg and 300 W Xe at >420 nm	H ₂ /O ₂ from pure water at rates of 16.6/8.3 μmol/h/0.5 g with QE = 0.66%; nondoped NiO–InTaO ₄ gave 3.2/1.1 μmol/h ^{47,62}
27–29	MTa ₂ O ₆ (M = Sr, Ba, Sn)	NiO, Pt	UV, 400 W Hg	NiO–SrTa ₂ O ₆ gives H ₂ /O ₂ from pure water at up to 960/490 μmol/h/g (QE = 7%); ^{147–149} BaTa ₂ O ₆ (orthorhombic) gives 33/15 μmol/h/g and 780/350 μmol/h/g with Ba(OH) ₂ and NiO cocat.; ⁴⁸ Pt/SnTa ₂ O ₆ makes 2.1 μmol/h/0.3 g of H ₂ under UV from MeOH(aq) ¹⁷⁹
30–34	MTa ₂ O ₆ (M = Ni, Mn, Co) and MTaO ₄ (M = Cr, Fe)		UV, 400 W Hg	M = Ni gave H ₂ /O ₂ at 11/4 μmol/h/g, M = Mn, Co, and CrTaO ₄ , FeTaO ₄ gave only traces of H ₂ ⁴⁸
35	Ca ₂ Ta ₂ O ₇	NiO	450 W Hg	170/83 μmol of H ₂ /O ₂ per 0.5 g from pure water ¹⁴⁸
36, 37	Bi ₂ MTaO ₇ (M = La, Y)		400 W Hg	Bi ₂ LaTaO ₇ was slightly more active with H ₂ /O ₂ at 41.8/20.5 μmol/h/g from pure water ¹⁵⁰
38	Sr ₂ Ta ₂ O ₇	NiO	UV, 400 W Hg	H ₂ /O ₂ from pure water at 2141/1059 μmol/h/0.5 g ^{149,151}
39–41	M ₂ La _{2/3} Ta ₂ O ₇ (M = K, H), H ₂ SrTa ₂ O ₇	NiO	UV, 400 W Hg	H ₂ /O ₂ from pure water at rates of 262/– (K ₂ La _{2/3} Ta ₂ O ₇), 940/459 (H ₂ La _{2/3} Ta ₂ O ₇), 240/59 μmol/h/0.5 g (H ₂ SrTa ₂ O ₇); without NiO, the rates are half, except for H ₂ SrTa ₂ O ₇ , which is more active (385/179 μmol/h/0.5 g) ¹⁴⁴
42	K ₂ Sr _{1.5} Ta ₃ O ₁₀	RuO ₂	400 W Hg	H ₂ /O ₂ from pure water at 100/39.4 μmol/h/0.5 g (QE = 2%); without cocat. activity reduced to 17% of these values ¹⁵²
43	KBa ₂ Ta ₃ O ₁₀	NiO	400 W Hg	150 μmol/h/g of H ₂ (QE = 8%) ¹²²
44–46	Sr ₄ Ta ₂ O ₉ , M ₃ Ta ₄ O ₁₅ (M = Sr, Ba)	NiO	400 W Hg	H ₂ /O ₂ from pure water at 32/2 μmol/h/0.5 g (Sr ₄ Ta ₂ O ₉) and 1194/722 μmol/h/0.5 g (Sr ₅ Ta ₄ O ₁₅); ¹⁴⁹ Ba ₅ Ta ₄ O ₁₅ gave up to 2.08/0.91 mmol/h/g of H ₂ /O ₂ and 7.1/3.6 mmol/h/g in the presence of Ba _{0.5} TaO ₃ traces ¹⁵³
47	K ₃ Ta ₃ Si ₂ O ₁₃	NiO	UV, 400 W Hg	H ₂ /O ₂ at 43/19 μmol/h/g; NiO improves activity to 368/188 μmol/h/g ¹⁵⁴
48	K ₃ Ta ₃ B ₂ O ₁₂		450 W Hg	0.5 g of cat. produced 2.4/1.2 mmol of H ₂ /O ₂ per h from pure water; addition of NiO did not improve the activity ¹⁵⁵
49–52	R ₃ TaO ₇ (R = Y, Yb, Gd, La)		400 W Hg	La ₃ TaO ₇ (Weberite), 164/82 μmol/h/0.5 g of H ₂ /O ₂ from pure water; Pyrochlore, 4/1 μmol/h/0.5 g; ^{123,156} for R = Y, Yb, only H ₂ traces
53–57	LnTaO ₄ (Ln = La, Ce, Pr, Nd, and Sm)	NiO	400 W Hg	only NiO-modified LaTaO ₄ has signif. cat. activity with H ₂ /O ₂ at rates of 115.6/51.5 μmol/h/0.2 g ¹⁵⁷
58–61	RbLnTa ₂ O ₇ (Ln = La, Pr, Nd, and Sm)		UV, 400 W Hg	up to 47/25.3 μmol/h/0.2 g of H ₂ /O ₂ (RbNdTa ₂ O ₇) from pure water; NiO cocat., 117.2/58.7 μmol/h/0.2 g; non-NiO-modified NaNdTa ₂ O ₇ , 2.4/<0.1 μmol/h/0.2 g ^{158,159}
62, 63a	M ₄ Nb ₆ O ₁₇ (M = K, Rb)		UV, 450 W Hg	M = K gives nonstoich. water splitting at a low rate; ¹⁶⁰ addition of NiO gives H ₂ /O ₂ at rates of 70/35 μmol/h/g; under opt. conditions, QE = 5.3–20%; ^{122,161–163} after internal platinization with [Pt(NH ₄) ₄] ²⁺ , 45 μmol/h/g of H ₂ with stoich. O ₂ , QE = 1.3%; ¹⁶⁴ for M = Rb, QE = 10% was observed with NiO ⁴³
63b	K ₄ Nb ₆ O ₁₇ –TiO ₂ intercalated		UV, 450 W Hg	1 g of this material in pure water at 60 °C evolved 1 mL of gases per h; H ₂ /O ₂ content was not determined ¹⁶⁵
63c	K ₄ Nb ₆ O ₁₇ , H ⁺ -exchanged	Pt	500 W Hg/Xe at >400 nm	0.2 μmol/h/0.1 g of H ₂ for K ₂ H ₂ Nb ₆ O ₁₇ (QE = 0.3%) from aqueous HI in the presence of [Ru(bpy) ₃] ²⁺ sensitizer ^{74,168}
64–66	KNb ₃ O ₈ , KTiNbO ₅ , CsTi ₂ NbO ₇	Pt	500 W Hg/Xe at >400 nm,	<0.2 μmol/h/0.1 g of H ₂ from aqueous HI with [Ru(bpy) ₃] ²⁺ sensitizer ^{74,168}
67a–d	A[M _{n–1} Nb _n O _{3n+1}] (A = H, K, Rb, Cs; M = Ca, Sr, Na, Pb; n = 2–4)	Pt	UV, 450 W Hg	Pt-mod. cat. produces up to 100 μmol/h/g of H ₂ but no O ₂ (value for KCa ₂ Nb ₃ O ₁₀) from MeOH(aq); HCa ₂ Nb ₃ O ₁₀ gives 5.9 mmol/h/g; modified with Pt increases the rate to 19 mmol/h/g; ^{38,43,169} 620 μmol/h/0.1 g for restacked Pt/HCa ₂ Nb ₃ O ₁₀ from MeOH(aq) (500 W Xe lamp); ¹⁷² 8.1 mmol/h/g of H ₂ for Pt/HCa ₂ Nb ₃ O ₁₀ /SiO ₂ from MeOH(aq) ¹⁷¹
67b	KCa ₂ Nb ₃ O ₁₀	RuO ₂	450 W Hg	H ₂ at 3.8 μmol/h/0.3 g from pure water but no O ₂ ; intercalation with RuOx gives H ₂ /O ₂ at 96/47 μmol/h/0.3 g ¹⁷⁰

Table 1. Continued

no.	material	cocatalyst	irradiation conditions	reaction details and refs
67c	HCa ₂ Nb ₃ O ₁₀	Pt	750 W Hg	individual nanosheets produce H ₂ at 2.28 $\mu\text{mol/h}$ /0.1 g from pure water (QE = 0.22%) and 78 $\mu\text{mol/h}$ /0.1 g after modification with Pt (QE = 7.5%); no O ₂ is produced ¹⁷³
68	K _{0.5} La _{0.25} Bi _{0.25} Ca _{0.75} Pb _{0.75} Nb ₃ O ₁₀	Pt	450 W Xe at >420 nm	traces of H ₂ and 168 $\mu\text{mol/h}$ /0.3 g of O ₂ from MeOH(aq)/AgNO ₃ ¹²⁷
69, 70	MPb ₂ Nb ₃ O ₁₀ (M = Rb, H)	Pt	500 W Xe at >420 nm	cat. with M = Rb evolves H ₂ traces from MeOH(aq); Pt-modified HPb ₂ Nb ₃ O ₁₀ gives ~ 4 $\mu\text{mol/h}$ /g and internally platinized cat. gives ~ 14.5 $\mu\text{mol/h}$ /g from MeOH(aq) ¹⁷⁴
71	PbBi ₂ Nb ₂ O ₉	Pt	450 W Xe at >420 nm	7.6 $\mu\text{mol/h}$ /g of H ₂ (QE = 0.95%) from MeOH(aq); using AgNO ₃ , it produces O ₂ at a rate of 520 $\mu\text{mol/h}$ /g (QE = 29%) ¹¹³
72	Bi ₃ TiNbO ₉	Pt	UV, 450 W Hg	33 $\mu\text{mol/h}$ /g of H ₂ from MeOH(aq) and 31 $\mu\text{mol/h}$ /g of O ₂ from aqueous AgNO ₃ ¹⁷⁵
73–77	Bi ₂ MNbO ₇ (M = Al, Ga, In), M ₂ BiNbO ₇ (M = Ga, In)		UV, 400 W Hg	Bi ₂ MNbO ₇ from H ₂ from MeOH(aq) and O ₂ from Ce(SO ₄) ₂ solution at rates up to 710 and 25 $\mu\text{mol/h}$ /g, respectively; cat. with M = Al was most active; ⁴⁷ M ₂ BiNbO ₇ evolve 54.3 (In) and 72.6 (Ga) $\mu\text{mol/h}$ /g of H ₂ with stoich. O ₂ ¹⁷⁷
78	ZnNb ₂ O ₆		UV, 400/450 W Hg	small amounts of H ₂ but no O ₂ ; with NiO water splitting with 54/21 $\mu\text{mol/h}$ /g of H ₂ /O ₂ ^{151,178}
79	SnNb ₂ O ₆	Pt	300 W Xe at >420 nm	evolves H ₂ at 18 $\mu\text{mol/h}$ /0.3 g from MeOH(aq) solution with a Pt cocat. ¹⁷⁹
80	Sr ₂ Nb ₂ O ₇		UV, 450 W Hg	NiO-loaded Sr ₂ Nb ₂ O ₇ produced H ₂ /O ₂ , at rates of 110/36 $\mu\text{mol/h}$ /g; without NiO, only H ₂ was produced; ¹⁵¹ larger activity (402/198 $\mu\text{mol/h}$ /g of H ₂ /O ₂ and QE = 23%) were observed by Lee and co-workers ¹²²
81	Ca ₂ Nb ₂ O ₇	NiO	UV, 450 W Hg	H ₂ at a rate of 101 $\mu\text{mol/h}$ /g with a QE = 7%; ¹²² O ₂ not determined
82	Ca ₂ Nb ₄ O ₁₁		UV, 400 W Hg	1.7/0.8 mmol/h/0.5 g of cat. ¹⁸⁰
83	Ba ₅ Nb ₄ O ₁₅	NiO	400 W Hg	from pure water H ₂ and O ₂ are evolved at 650/250 $\mu\text{mol/h}$ /0.5 g (solid-state reaction) and 2366/1139 $\mu\text{mol/h}$ /0.5 g (preparation from metal citrates) ¹⁸¹
84	BiVO ₄		300 W Xe at >520 nm	O ₂ at 31 $\mu\text{mol/h}$ /g from aqueous AgNO ₃ (QE = 0.5%) at 450 nm ¹⁸²
85	Ag ₃ VO ₄		300 W Xe at >420 nm	makes 17 $\mu\text{mol/h}$ /0.3 g of O ₂ from an aqueous AgNO ₃ solution ¹⁸³
86a	WO ₃		500 W Xe at >330 nm	O ₂ rate (1220 $\mu\text{mol/h}$ /g) in the presence of Fe ³⁺ or Ag ⁺ ; ¹⁸⁴ lower O ₂ rates (38 $\mu\text{mol/h}$ /0.8 g in the first hour) were measured by Arakawa and co-workers with FeCl ₃ as the oxidizer together with traces (<0.7 $\mu\text{mol}/10$ h/0.8 g) of H ₂ ¹⁸⁵
86b	WO ₃	Pt	300 W Xe at >420 nm	Pt–WO ₃ alone with NaIO ₃ produced O ₂ at 84 $\mu\text{mol/h}$ /0.2 g but no H ₂ ; combined with Pt/SrTiO ₃ , H ₂ /O ₂ evolved under vis light (420.7 nm) at rates of 0.21 and 0.11 $\mu\text{mol/h}$ /0.4 g of both cats. (QE = 0.1%); ^{118,186} with TaON instead of SrTiO ₃ , H ₂ /O ₂ rates of 24/12 $\mu\text{mol/h}$ /0.4 g, QE = 0.4% at 420 nm ¹⁸⁷
87, 88	Na ₂ W ₄ O ₁₃ , Si(W ₃ O ₁₀) ₄] ^{4–}		UV, 400 W Hg	21/9 $\mu\text{mol/h}$ /g of H ₂ /O ₂ from aqueous MeOH or Ag ⁺ ; in comparison, the protonated polytungstate ion Si(W ₃ O ₁₀) ₄] ^{4–} with four H ⁺ produced H ₂ (156 $\mu\text{mol/h}$ /g) with methanol but no O ₂ ¹⁸⁴
89	Ca ₂ NiWO ₆	Pt	UV, 300 W Xe	H ₂ at 4.12 $\mu\text{mol/h}$ /0.5 g from aqueous MeOH; from aqueous AgNO ₃ , O ₂ is evolved at 0.38 $\mu\text{mol/h}$ /0.5 g (0.36 $\mu\text{mol/h}$ /0.5 g for vis light irradiation) ¹⁸⁸
90, 91	Bi ₂ W ₂ O ₉ , Bi ₂ WO ₆	Pt	UV, 450 W Hg	H ₂ from MeOH(aq) at rates of 18 (Bi ₂ W ₂ O ₉) and 1.6 (Bi ₂ WO ₆) $\mu\text{mol/h}$ /g; Bi ₂ WO ₆ evolves 3 $\mu\text{mol/h}$ /g of O ₂ with vis light from an aqueous AgNO ₃ solution ¹⁷⁵
92–94	Bi ₂ MoO ₆ , Bi ₂ Mo ₂ O ₉ , Bi ₂ Mo ₃ O ₁₂		UV, 300 W Xe at >300 or >420 nm	all evolve O ₂ from AgNO ₃ (aq) but no H ₂ even from MeOH(aq); UV, low-temp (LT) modification of Bi ₂ MoO ₆ (127 $\mu\text{mol/h}$ /0.5 g), Bi ₂ Mo ₃ O ₁₂ (46 $\mu\text{mol/h}$ /0.5 g), Bi ₂ Mo ₂ O ₉ (1.8 $\mu\text{mol/h}$ /0.5 g), high temp modification of Bi ₂ MoO ₆ (0.7 $\mu\text{mol/h}$ /0.5 g); vis light, LT Bi ₂ MoO ₆ (55 $\mu\text{mol/h}$ /0.5 g), Bi ₂ Mo ₃ O ₁₂ (7.6 $\mu\text{mol/h}$ /0.5 g) ¹⁸⁹
95a/b	PbMoO ₄ , Cr:PbMoO ₄		200 W Hg and 300 W Xe at >300 or >420 nm	from MeOH(aq), Pt/PbMoO ₄ makes 59 $\mu\text{mol/h}$ /0.3 g of H ₂ under UV (Hg) and 3 $\mu\text{mol/h}$ /0.3 g of O ₂ , but no H ₂ , from pure water; AgNO ₃ (aq), PbMoO ₄ evolves 97 $\mu\text{mol/h}$ /0.35 g of O ₂ under UV (Hg); ¹⁹⁰ under Xe UV light, the Cr-doped phases make up to 120 $\mu\text{mol/h}$ /0.5 g of O ₂ and up to 71.5 $\mu\text{mol/h}$ /0.5 g of O ₂ under vis light ¹⁹¹
96	CeO ₂		500 W Xe at >330 nm	using Fe ₂ (SO ₄) ₃ as the oxidizer, O ₂ at ~ 2.5 $\mu\text{mol/h}$ /0.8 g of cat. ¹⁸⁵

Table 1. Continued

no.	material	cocatalyst	irradiation conditions	reaction details and refs
97	ZrO ₂		UV, 400 W Hg	72/36 $\mu\text{mol/h/g}$ of H ₂ /O ₂ from pure water; addition of NaHCO ₃ increases the rate (309/167 $\mu\text{mol/h/g}$); ¹³⁷ Pt, Cu, Au, or RuO ₂ cocats. reduce the activity
98	Cu ₂ O		300 W Xe at >460 nm	H ₂ /O ₂ at rates of 1.7/0.9 $\mu\text{mol/h/0.5 g}$ of cat.; ¹⁹² activity partly due to mechanocatalysis ¹⁹³
99a/b	In ₂ O ₃ /Cr:In ₂ O ₃	NiO, Pt	UV, 400 nm, Hg, or >420 nm, Xe	vis, 0.36/1.30 $\mu\text{mol/h/0.5 g}$ of H ₂ /O ₂ from MeOH(aq)/AgNO ₃ ; UV, 1.1 $\mu\text{mol/h/0.5 g}$ from pure water and no O ₂ ; ¹⁹⁴ for Cr:In ₂ O ₃ , the rates are lower
100a/b	In ₂ O ₃ (ZnO)(<i>m</i>) (<i>m</i> = 3, 9)		300 W Xe at >420 nm	for <i>m</i> = 3, 1.1/1.3 $\mu\text{mol/h/g}$ of H ₂ /O ₂ from MeOH(aq)/AgNO ₃ solution; ¹⁹⁶ lower rates for <i>m</i> = 9
101	Ba ₂ In ₂ O ₅	NiO, Pt	UV, 400 nm, Hg, or >420 nm, Xe	vis, 3.2/0.46 $\mu\text{mol/h/0.5 g}$ of H ₂ /O ₂ from MeOH(aq)/AgNO ₃ ; UV, 4.2 $\mu\text{mol/h/0.5 g}$ from pure water and no O ₂ ; for Cr:Ba ₂ In ₂ O ₅ , the rates are lower; for In ₂ O ₃ –Ba ₂ In ₂ O ₅ , 7.8 $\mu\text{mol/h/0.5 g}$ of H ₂ from pure water (UV); ¹⁹⁴ Cr doping gives 7.9/0.35 $\mu\text{mol/h/0.5 g}$ of H ₂ /O ₂ from MeOH(aq)/AgNO ₃ (vis) and 29.3/15.2 $\mu\text{mol/h/0.5 g}$ from pure water (UV)
102–104	MIn ₂ O ₄ (M = Ca, Sr), LaInO ₃	RuO ₂	UV, 400 W Xe (280–700 nm) or 200 W Hg/Xe (230–436 nm)	RuO ₂ –CaIn ₂ O ₄ , H ₂ /O ₂ at up to ~21/10 $\mu\text{mol/h/0.25 g}$ of cat.; RuO ₂ –SrIn ₂ O ₄ , H ₂ /O ₂ at rates of ~7/3.5 $\mu\text{mol/h/0.25 g}$; RuO ₂ –LaInO ₃ , ~1/0.5 $\mu\text{mol/h/0.25 g}$; ¹⁹⁷ deviating yields in other references ^{200,260}
105	Zn:Lu ₂ O ₃ /Ga ₂ O ₃	NiO	UV, 400 W Hg	50.2 $\mu\text{mol/h/0.5 g}$ for H ₂ and 26.7 $\mu\text{mol/h/0.5 g}$ for O ₂ , respectively, with QE = 6.81% at 320 nm ¹⁹⁸
106	ZnGa ₂ O ₄	RuO ₂	UV, 200 W Hg/Xe	H ₂ and O ₂ from pure water with rates of ~10/~3.5 $\mu\text{mol/h/0.25 g}$ of cat. ¹⁹⁹
107	Sr ₂ SnO ₄	RuO ₂	UV, 400 W Hg	H ₂ /O ₂ evol. at 13/5 $\mu\text{mol/h/0.25 g}$ from pure water ²⁰⁰
108–111	M ₂ Sb ₂ O ₇ (M = Ca, Sr), CaSb ₂ O ₆ , NaSbO ₃	RuO ₂	200 W Hg/Xe	H ₂ (near stoich. O ₂) at 2.8 (Ca ₂ Sb ₂ O ₇), 7.8 (Sr ₂ Sb ₂ O ₇), 1.5 (CaSb ₂ O ₆), and 1.8 (NaSbO ₃) $\mu\text{mol/h/0.25 g}$ of cat.; no H ₂ without RuO ₂ ; ²⁰¹ an earlier paper reports H ₂ /O ₂ rates of 4.5/2.5 $\mu\text{mol/h/0.25 g}$ for NaSbO ₃ ²⁰⁰
112	Zn ₂ GeO ₄	RuO ₂	Hg/Xe (power not specified)	H ₂ (22 $\mu\text{mol/h}$) and O ₂ (10 $\mu\text{mol/h}$) from pure water (amount of cat. not specified) ²⁰²
113	GaN:ZnO	RuO ₂ or Cr/Rh oxide	UV, 450 W Hg at >300 nm and 300 W Xe at >400 nm	at pH = 3, H ₂ /O ₂ at ~1/0.29 mmol/h/0.3 g; QE for O ₂ evol. = 0.14% (for 300–480 nm) to 0.23% (for >420 nm) ^{203,204} with RuO ₂ cocat.; with Cr/Rh cocat., QE = 2.5% at 420–440 nm ^{14,15,205}
114	β -Ge ₃ N ₄	RuO ₂	UV, 450 W Hg	stoich. H ₂ /O ₂ at H ₂ rates of ~0.5 mmol/h/0.5 g, QE = 9% at 300 nm; pure Ge ₃ N ₄ is inactive ^{206,207}
115	Zn _{1.44} GeN _{2.08} O _{0.38}	RuO ₂	UV, 450 W Hg at >400 nm	UV, H ₂ /O ₂ rates of 54.3/27.5 $\mu\text{mol/h/0.2 g}$ of cat.; no activity without RuO ₂ ; under vis light (>400 nm), cat. evolves H ₂ /O ₂ at 14.2/7.4 $\mu\text{mol/h/0.2 g}$; ²⁰⁸ ZnO or ZnGeN ₂ alone do not split water with or without RuO ₂ cocat.
116	Ta ₃ N ₅	Pt	300 W Xe at >420 nm	H ₂ at 1.8 $\mu\text{mol/h/0.2 g}$ from MeOH(aq) (QE = 0.1% for 420–600 nm); O ₂ at up to 100 $\mu\text{mol/h/0.2 g}$ from AgNO ₃ (aq) with La ₂ O ₃ buffer (pH = 8.5) ^{69,209}
117a	TaON	Pt or Ru	300 W Xe at >420 nm	O ₂ at a rate of 380 $\mu\text{mol/h/0.4 g}$ (QE = 34%) during the first hour from AgNO ₃ (aq) with La ₂ O ₃ as a base buffer (pH = 8); ²¹⁰ H ₂ at 4 $\mu\text{mol/h}$ (QE = 0.2%) with 0.4 g of Pt–TaON from MeOH(aq); Ru–TaON, 120 $\mu\text{mol/h/0.4 g}$ of H ₂ (QE = 0.8%) ^{44,211}
117b	TaON		300 W Xe at >420 nm	H ₂ /O ₂ rates of 24/12 $\mu\text{mol/h/0.4 g}$ (0.2 g of each cat.) in tandem with Pt–WO ₃ and iodide as a redox shuttle (QE = 0.4% at 420 nm) ¹⁸⁷
118–120	MTaO ₂ N (M = Ca, Sr, Ba)		300 W Xe at >420 nm	H ₂ forms at 15–20 $\mu\text{mol/h/0.2 g}$ from MeOH(aq); no O ₂ from AgNO ₃ (aq) ⁴⁴
121	LaTiO ₂ N	Pt or IrO ₂	300 W Xe at >420 nm	for Pt cocat., H ₂ /O ₂ at ~8/29 $\mu\text{mol/h/0.2 g}$ from MeOH(aq)/AgNO ₃ (with La ₂ O ₃ buffer for pH = 8) during the first 10 h; for H ₂ , QE = 0.15%; for IrO ₂ as cocat., 200 $\mu\text{mol/h/0.25 g}$ of O ₂ (QE = 5%) ²¹²
122	Y ₂ Ta ₂ O ₅ N ₂		300 W Xe at >420 nm	0.3 g of Pt/Ru-modified cat. evolves H ₂ at 250 $\mu\text{mol/h}$ from aqueous ethanol; the O ₂ rate was 140 $\mu\text{mol/h/0.3 g}$ from AgNO ₃ (aq) and La ₂ O ₃ as a base buffer (pH = 8) ²¹³
123	InP	Pt	UV, 250 W Hg	H ₂ at 2–5 $\mu\text{mol/h/30 mg}$ of cat. from SO ₃ ^{2–} (aq) or S ^{2–} ; cat. was not characterized ²¹⁴
124a	CdS	Pt	150 W Hg at >360 nm	1–10 $\mu\text{mol/h}$ of H ₂ for 5 mg of cat. with EDTA as the sacrif. donor; decomp. after >4 h ^{93,218,219}
124b	CdS, doped with ZnS and Ag ₂ S	Pt	900 W Xe at >300 nm or sunlight	Pt–CdS, up to 66 mL/h/0.4 g of cat. (=2.9 mmol/h), QE = 25% from aqueous S ^{2–} , SO ₃ ^{2–} , or S ^{2–} /HPO ₄ ^{217,220}
124c	CdS	Pt/RuO ₂	450 W Xe at >400 nm	QE = 37% for 15 mol % ZnS:CdS ²¹⁷ 2.8 mL of H ₂ (125 μmol) and 1.4 mL of O ₂ per 44 h per 2.75 mg of cat. from pure water; ^{221,222} O ₂ evol. was questioned ^{21,70,93}

Table 1. Continued

no.	material	cocatalyst	irradiation conditions	reaction details and refs
124d	CdS	variable	variable	micelles, ^{223–225} CdS composites with (TiO ₂ , ²²⁶ ZnS, ^{228,229} CdSe ²³⁰) different cocats. (Pt, Pd, Rh, Ru, Ir, Fe, Ni, Co), ^{231,232} hollow CdS microparticles, ²³³ Cu-doped CdS, ^{234,235} effect of preparation ^{236,237}
125a	ZnS		125 W Hg at >290 nm	0.5 mL of H ₂ /h (22 μmol) per 12 mg of cat. from THF(aq) ²³⁸
125b	ZnS	Pt	200 W Hg at >300 nm	295 mL/h (13 mmol/h) of H ₂ from 0.4 g of cat., QE = 90%; ¹³ from aqueous S ^{2–} , SO ₃ ^{2–} , S ^{2–} /HPO ₄ ^{2–} , or S ₂ O ₃ ^{2–} solution
125c	M:ZnS (M = Ni, Pb, Cu)		300 W Xe at >420 nm	M = Ni, 280 μmol/h/g (QE = 1.3%) of H ₂ from aqueous K ₂ SO ₃ and Na ₂ S; ²⁴⁰ M = Pb, 15 μmol/h/g from aqueous SO ₃ ^{2–} ; ²⁴¹ M = Cu, 450 μmol/h/g of H ₂ were evolved (QE = 3.7%) from Na ₂ SO ₃ (aq) ²³⁹
125d	ZnS, doped with AgInS ₂ or CuInS ₂	Pt or Ru	300 W Xe at >420 nm	for Ru-modified (CuAg) _{0.15} In _{0.3} Zn _{1.4} S ₂ , H ₂ at up to 2.3 mmol/h/0.3 g from aqueous Na ₂ S and K ₂ SO ₃ with QE = 7.5%; ²⁴³ Pt-loaded (AgIn) _{0.22} Zn _{1.56} S ₂ has QE = 20% ²⁴⁵
126–128	Na ₁₄ In ₁₇ Cu ₃ S ₃₅ , (AEP) ₆ In ₁₀ S ₁₈ , Na ₅ In ₇ S ₁₃		300 W Xe at >420 nm	9 μmol/h/0.5 g of H ₂ from Na ₂ S(aq) with QE = 3.7% (QE = 0.37% for SO ₃ ^{2–} (aq)); (AEP) ₆ In ₁₀ S ₁₈ evolved 20 μmol/h/0.5 g; ²⁴⁷ AEP = protonated 1-(2-aminoethyl)piperazine; Na ₅ In ₇ S ₁₃ produced 2.4 μmol/h/0.25 g from Na ₂ SO ₃ (aq) ²⁴⁶
129a/b	[In(OH) _y S _z] with or without Zn doping		300 W Xe at >420 nm	0.9–1.8 μmol/h/0.3 g from aqueous Na ₂ S/Na ₂ SO ₃ ; for Pt/In(OH) _y S _z :Zn, activity up to 67 μmol/h/0.3 g (QE = 0.59%) under similar condions ²⁴⁸
130–132	MInS ₂ (M = Cu, Na, CuInS ₈)		UV, 400 W Xe	CuInS ₂ (0.006 mL/h/0.5 g of cat.) and CuInS ₈ (0.04 mL/h/0.5 g of cat.) from Na ₂ SO ₃ (aq); ²⁴⁹ NaInS ₂ –Pt produces H ₂ (470 μmol/h/0.7 g) from K ₂ SO ₃ (aq); ²⁵⁰ under vis light (>420 nm, 300 W Xe).
133	WS ₂		1000 W Xe at >435 nm	0.05 mL of H ₂ /h/10 mg of cat. from EDTA(aq) with fluoresceine as the sensitizer and SiO ₂ as the support ²⁵¹
134	Bi ₂ S ₃		500 W halogen lamp, vis	H ₂ at 0.011 mL/h/0.001 g from aqueous sulfide; platinization improves the activity by 25% ²⁵²

catalytic activity of these systems was found to strongly depend on the doping levels of the catalysts, the TiO₂ concentration, the pH, and the temperature.⁹⁰ Later it was discovered that O₂ production was less than stoichiometric²⁹ or entirely absent in some cases.^{21,93} This was attributed to the fact that photoreduced TiO₂ tends to strongly adsorb oxygen as O₂[–]^{29,93} or as O₂^{2–}.⁹⁴ Following initial reports on the beneficial effect of a NaOH coating on the photocatalytic efficiency,⁹⁵ Arakawa's group reported in 1992 that the O₂ evolution activity of TiO₂⁹⁶ could be enhanced significantly by the addition of 0.1–2.2 M Na₂CO₃ to the aqueous catalyst dispersion. A similar effect was observed later for Ta₂O₅, ZrO₂, SrTiO₃, K₄Nb₆O₁₇, Na₂Ti₂O₁₃, and BaTi₄O₉.^{40,41} On the basis of IR spectroscopy, the authors hypothesize that surface-adsorbed peroxycarbonate species are involved in facilitating O₂ release.⁴¹ While Na₂CO₃ was found to also prevent the back-recombination of H₂ and O₂,⁴¹ an alternative way to suppress this reaction is to substitute Pt in TiO₂/Pt catalysts with Pd and Rh.⁹⁵ The rate [h^{–1}] of the back-recombination reaction decreased in the order of Pd (0.23–0.51) > Pt (0.32–1.8) > Rh (0.20–0.30); however, Rh was also the least active cocatalyst for the forward reaction.⁹⁵ For certain niobates and tantalates (in particular, La-doped NaTaO₃), it was recently shown that H₂/O₂ back-recombination can be fully prevented with Au nanoparticles as cocatalysts.⁹⁷ However, for Au the reduction of O₂ takes place in competition with H₂O.

Unfortunately, the mechanism of photochemical reactions on the surface of TiO₂ has not yet been fully understood^{78,98} and is subject to ongoing studies.^{99–102} It is known though that irradiation with UV light produces electron–hole pairs,

which become trapped after 250 ns (holes) and 20 ps (electrons) to produce absorptions in the visible at 475 nm (holes) and 650 nm (electrons), respectively.^{103–105} Electrons can be trapped as Ti³⁺ ions,^{105,106} which have been shown to be capable of reducing water to H₂.⁷⁸ Holes are trapped on surface hydroxyl sites,^{81,104} which are believed to react with water to eventually produce O₂.

More recent work on TiO₂ water-splitting catalysts includes tandem systems with separate semiconductors for water reduction and oxidation (Figure 1D).¹⁰⁷ Under UV irradiation, Pt–TiO₂ (Anatase) was found to preferentially catalyze water reduction with iodide as the sacrificial electron donor, whereas TiO₂ (Rutile) was found to be the superior water oxidation catalyst in the presence of IO₃ as the electron acceptor. After both catalysts were combined, H₂ and O₂ were formed stoichiometrically from a basic (pH = 11) solution, with iodide serving as a redox shuttle. A similar system was realized with Pt–TiO₂ (Anatase) and Pt–WO₃, giving QE = 4% upon irradiation with UV light.¹⁰⁸ Other recent efforts have sought to improve the optical response of TiO₂-based catalysts via doping with C¹⁰⁹ or N^{110,111} and S.¹¹² Of these materials, only N-doped TiO₂ has been tested for photocatalytic water splitting. Under visible light, the Pt-modified catalyst evolves O₂ from aqueous AgNO₃ as the sacrificial electron acceptor and traces of H₂ from aqueous methanol as the sacrificial electron donor.¹¹³

Titanates

When TiO₂ is fused with metal oxides (SrO, PbO, etc.), metal titanates with intermediate band gaps can be obtained.

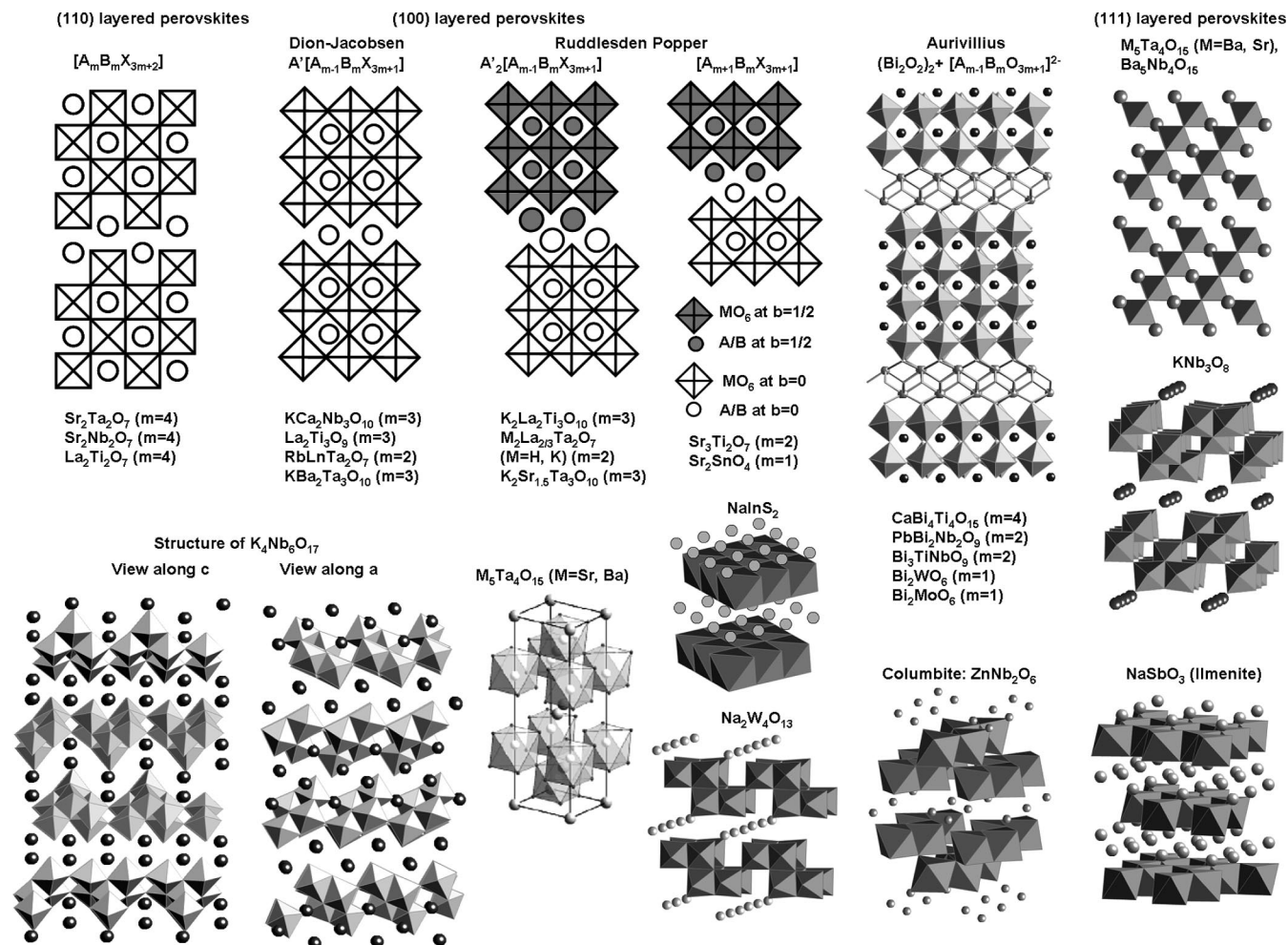


Figure 2. Crystal structures of layered semiconductors. For general structures ABX, A atoms are represented as spheres and BX units are shown as polyhedra.

Of these, $SrTiO_3$ crystallizes in the Perovskite structure type and has a band gap of 3.2 eV, slightly larger than that of TiO_2 . It was first employed in 1976 as a photocatalyst in a water-splitting electrochemical cell together with p-CdTe or p-GaP photocathodes.¹¹⁴ An optimized version of this system was shown in 1977 by Ohashi and co-workers to be the first self-supported photoelectrochemical cell¹¹⁵ with a photon-to-electron conversion efficiency of 0.044–0.67%. In 1980, it was shown that NiO-modified $SrTiO_3$ powder splits water vapor stoichiometrically under UV irradiation, while $SrTiO_3$ alone did not show any activity.¹¹⁶ The conditions of the NiO deposition had a strong effect on the catalytic activity.¹¹⁷ Abe et al. used Cr/Ta-doped Pt/ $SrTiO_3$ together with Pt– WO_3 in a two-particle catalyst system (z scheme, Figure 1D) for overall water splitting with visible light. Here Cr/Ta: $SrTiO_3$ –Pt produced H_2 and Pt– WO_3 produced O_2 , with an iodide/iodate redox couple serving as the redox mediator between the two catalysts.¹¹⁸ The effect of metal cocatalysts (Ru, Ir, Pd, Pt, Os, Re, Co) on the water-splitting activity of $SrTiO_3$ was studied by Lehn et al.^{119,120} From pure water, H_2 and O_2 were evolved stoichiometrically, with the activity decreasing in the order $Rh > Ru > Re > Pt > Ir > Pd > Os > Co$. $SrTiO_3$ alone produced H_2 but no O_2 . Lee and co-workers investigated a series of layered Perovskites as photochemical water-splitting catalysts.^{121,122} $La_2Ti_2O_7$ is a member of the (110)-layered Perovskites $[A_m B_m X_{3m+2}]$ with

$m = 4$ (Figure 2); i.e., it consists of four TiO_6 unit thick slabs that are separated by layers of La^{3+} ions. Because of its large band gap of 3.8 eV,¹²³ it requires UV irradiation for catalytic activity. The NiO-modified catalyst is active for H_2/O_2 evolution with a QE up to 12%.^{121–123} By doping with BaO and addition of NaOH to the catalyst suspension, this activity could be further increased to QE = 50%,¹²¹ only slightly below that of the best catalyst, La-doped $NaTaO_3$ (56%, see below).¹² With Pt as a cocatalyst, the activity is substantially lower.¹²⁴ A hydrothermally synthesized material with larger surface area exhibits intermediate activity.¹²⁵ Doping of $La_2Ti_2O_7$ with Cr or Fe does not increase the H_2 activity under UV irradiation but allows H_2 production ($< 15 \mu\text{mol/h}$ for Pt–Cr: $La_2Ti_2O_7$) from aqueous methanol with visible light.¹²⁴ $La_2Ti_3O_9$ is a (100)-layered Perovskite with $m = 3$ (Figure 2). Under UV irradiation, the NiO-modified catalyst splits water at significantly lower rates than $La_2Ti_2O_7$.¹²¹ The activity of $La_2Ti_5O_{15}$ is slightly higher, even though the material does not form a layered structure. Instead, it contains corner-shared chains of TiO_5 pyramidal units (Figure 3).¹²¹ $Sr_3Ti_2O_7$ is a (100)-layered Perovskite of the Ruddlesden–Popper series $[A_{m+1} B_m X_{3m+1}]$ with $m = 2$ (Figure 2). From its absorption edge at 395 nm, its band gap can be estimated as 3.2 eV. Under UV irradiation, it produces only H_2 from water, but stoichiometric O_2 evolution can be achieved after modification with a NiO cocatalyst.

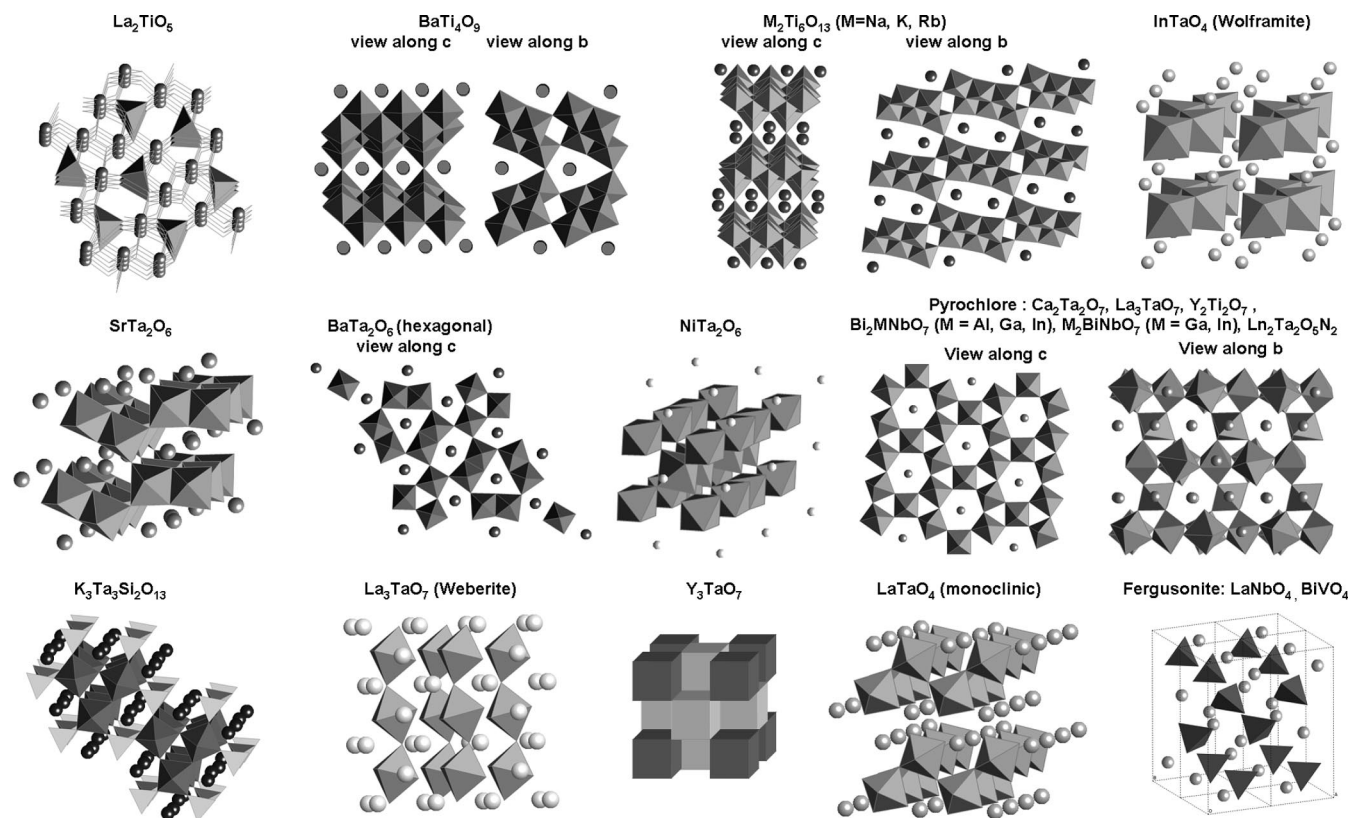


Figure 3. Crystal structures of semiconductors (part I). For general structures ABX, A atoms are represented as spheres and BX units are shown as polyhedra.

An alternate synthetic route produces a high-surface-area material with increased activity, after modification with NiO.¹²⁶ Doping of $\text{Sr}_3\text{Ti}_2\text{O}_7$ with Pb^{2+} cations leads to PbTiO_3 (Perovskite structure, not shown) with a reduced band gap of 2.98 eV.¹²⁷ After modification with Pt nanoparticles, visible light activates the catalyst to produce H_2 from aqueous methanol and O_2 from aqueous AgNO_3 . Doping of $\text{Sm}_2\text{Ti}_2\text{O}_7$ with sulfur produces $\text{Sm}_2\text{Ti}_2\text{S}_2\text{O}_5$, which belongs to the Ruddlesden–Popper-type (100)-layered Perovskites with $m = 2$ and with S occupying O sites. The semiconductor has a reduced band gap of 2.0 eV, from optical measurements. Under visible light irradiation, the Pt-modified catalyst produces H_2 in the presence of Na_2S – Na_2SO_3 or methanol but not from pure water. After modification with IrO_2 and $\text{Ca}(\text{OH})_2$, O_2 evolution proceeds in the presence of 0.01 M of the oxidizing agent AgNO_3 .¹²⁸ After NiO functionalization and doping of up to one Nb^{5+} on the Ti^{4+} sites, the Ruddlesden–Popper series ($m = 3$) $\text{M}_2\text{La}_2\text{Ti}_3\text{O}_{10}$ ($M = \text{K}, \text{Rb}, \text{Cs}$) cleaves water into H_2 and O_2 with QEs of up to 5%.^{129,130} The activity of $\text{K}_2\text{La}_2\text{Ti}_3\text{O}_{10}$ (shows photoluminescence at 475 nm at 77 K)¹³¹ is strongly dependent on the cocatalyst (Ni, Pt, or RuO_2 , Au¹³²) and on the concentration of added KOH.¹³³ It can be doubled by synthesizing the catalyst under conditions that produce submicrometer crystals.¹³⁴ As a member of the Aurivillius-type layered Perovskites ($m = 4$), $\text{CaBi}_4\text{Ti}_4\text{O}_{15}$ consists of quadruple-layer TiO_6 sheets separated by layers of $[\text{Bi}_2\text{O}_2]^{2+}$ (Figure 2). Substitution of Ca^{2+} with Pb^{2+} reduces the band gap of the material from 3.36 to 3.02 eV. Under visible light, the resulting $\text{PbBi}_4\text{Ti}_4\text{O}_{15}$ evolves small quantities of H_2 and O_2 from aqueous

MeOH and AgNO_3 , respectively.¹²⁷ Inoue studied the photocatalytic properties of a series of titanates (BaTi_4O_9 , $\text{Ba}_4\text{Ti}_{13}\text{O}_{30}$, $\text{Ba}_2\text{Ti}_9\text{O}_{20}$, $\text{Ba}_6\text{Ti}_{17}\text{O}_{40}$), whose structures contain tunnels occupied by alkaline-earth metal ions.⁴² Upon UV irradiation, only BaTi_4O_9 (Figure 3) was able to split water stoichiometrically into H_2 and O_2 after modification with RuO_2 . The activity of this catalyst was attributed to the presence of distorted TiO_6 octahedra whose stronger dipole moments aided electron–hole separation. In the related series $\text{M}_2\text{Ti}_6\text{O}_{13}$ ($M = \text{Na}, \text{K}, \text{Rb}$) (Figure 3) the tunnels are twice as large. After modification with RuO_2 particles, all members of the series produced H_2/O_2 from water, with the activity increasing with decreasing size of the cation.^{42,135} This trend was correlated with the sizes of the dipole moment in these structures originating from distorted TiO_6 octahedra. An additional factor was that <1 nm RuO_2 particles were thought to be able to make good electrical contact with the titanates, by fitting into the opening of the tunnel voids. $\text{La}_4\text{CaTi}_5\text{O}_{17}$ is a member of the 110-layered Perovskites $[\text{A}_m\text{B}_m\text{O}_{3m+2}]$ with $m = 5$ (Figure 2) and has a band gap of 3.8 eV. After modification with NiO, this catalyst was active for an overall water splitting under UV light with a QE of up to 20%.¹²²

Noncrystalline Ti/Zr phosphates were also explored as catalysts for photochemical water splitting. The polymerization of titanium and zirconium tetraisopropanolates in aqueous H_3PO_4 in the presence of octadecyltrimethylammonium chloride produces a series of amorphous Ti/Zr phosphates, which after modification with Pt metal evolve H_2 from water under UV light.¹³⁶ The activity increases with the Zr content to reach a maximum at a Ti/Zr molar ratio of 1:1.

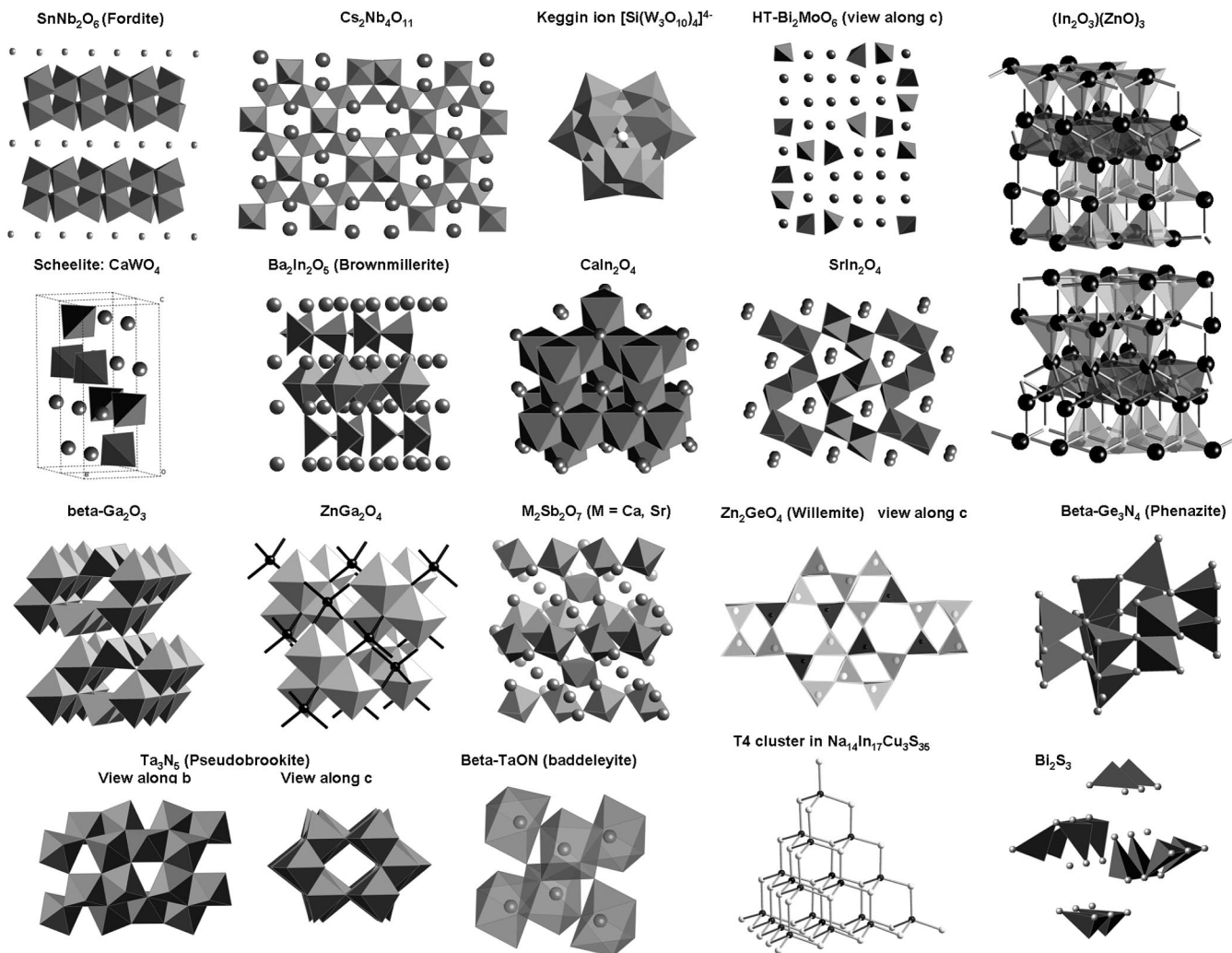


Figure 4. Crystal structures of semiconductors (part II). Continued from Figure 3.

Zirconium Oxide

ZrO₂ crystallizes in the Rutile structure type, like TiO₂, but its band gap of 5.0–5.7 eV^{137,138} is much larger. Under UV irradiation, pure ZrO₂ is an active catalyst for overall water splitting. The rate can be optimized by the addition of alkali carbonates to the water, of which NaHCO₃ works best.¹³⁷ Interestingly, the addition of Pt, Cu, Au, or RuO₂ as cocatalysts reduces the activity below the value observed in pure water. This is explained with the large bandgap and the positions of the conductance band and valence bands respectively, which lead to a large barrier height of the metal semiconductor junctions.

Tantalum Oxide and Tantalates

Tantalum oxide and the tantalates form another large group of water splitting catalysts. Because of their large bandgaps UV irradiation is needed for activity, and in many cases, metal or metal oxide cocatalysts. The bandgap of Ta₂O₅ (3.9–4.0 eV) lies between that of ZrO₂ (5.0 eV), and TiO₂ (3.0 eV). By itself it produces only traces of H₂ and no O₂ in pure water upon UV illumination. However, the addition of NiO, Pt, or RuO₂ cocatalysts converts the material into an active water-splitting catalyst (see Table 1).^{40,48} The crystal structure of Ta₂O₅ is not known, but first-principle

calculations suggest a hexagonal structure containing hexa- and octacoordinate Ta ions.¹³⁹ The first layered tantalates A₄Ta_xNb_{6-x}O₁₇ (A = Rb, K; x = 1–4) were synthesized by doping K₄Nb₆O₁₇ with Ta₂O₅ by Domen et al. in 1996.¹⁴⁰ These compounds are isomorphous with K₄Nb₆O₁₇ (see Figure 2). The band gaps in the series increase with the Ta content from 3.4 eV for Rb₄Nb₆O₁₇ to 4.2 eV for Rb₄Ta₆O₁₇, except for K₄Ta₄Nb₂O₁₇ whose absorption edge at 410 nm revealed a band gap of only 3.0 eV. After modification with NiO, all materials split water under UV irradiation, but the activity decreases with increasing Ta content. Kato and Kudo discovered in 1998 that tantalates MTaO₃ (M = Li, Na, K) are also very effective photocatalysts for water splitting under UV irradiation.^{48,141} The oxides crystallize in the Perovskite structure type, and their band gaps depend strongly on the cations, 4.7 eV (Li), 4.0 eV (Na), and 3.7 eV (K), as determined from diffuse-reflectance spectra.¹⁴² In combination with NiO as the cocatalyst, NaTaO₃ produced H₂ and O₂ from pure water with quantum yields of 20–28%.^{141,142} Without cocatalysts, the rates for H₂/O₂ production are 10–40 times lower, depending on the experimental conditions.¹⁴² Recent work by Teng and co-workers showed that NaTaO₃ produced by a sol–gel method showed higher activity for water splitting than the same material prepared by a high-temperature (HT) solid-state synthesis.¹⁴³ This was attributed

to the higher surface area of the sol–gel product and to its monoclinic crystal structure (instead of orthorhombic) with a slightly larger band gap (4.1 eV). In 2003, Kudo and co-workers reported that NaTaO_3 , doped with 2 mol % La and modified with a NiO cocatalyst, split pure water with a quantum yield of 56% under UV irradiation.¹² This is the highest QE ever reported for a water-splitting catalyst. The high performance was attributed to the La dopant, which reduced the catalyst particle size and caused the formation of nanosteps on the particle surface. According to the authors, the grooves in the nanosteps served as O_2 formation sites, while NiO particles at the step edges catalyzed H_2 formation. The related $\text{La}_{1/3}\text{TaO}_3$ (4.0 eV), on the other hand, has significantly lower activity with a NiO cocatalyst.¹⁴⁴ Doping of KTaO_3 with Ti, Zr, and Hf also increases its water-splitting activity but not above that of NiO–La:NaTaO₃.¹⁴⁵ Ti works best (at 8% doping level), followed by Zr and Hf. The cation size of the dopant could be correlated with the distortion of the structure and the O_2 production rate and inversely with the H_2 production rate. Zr:NaTaO₃ crystals coated with metal porphyrinoids and with Pt particles were found by Ishihara and co-workers to catalyze water splitting under UV irradiation with QEs of up to 12.8%.¹⁴⁶ The photocatalytic activity depended strongly on the nature of the porphyrin ligand and also on the metal cation (Zn, Co, Cr, Mg). Time-resolved measurements of the photovoltaic potential showed that the porphyrin prolongs the half-life of the excited state from 4.0 to 11.2 μs . It was suggested that the porphyrin complex aids the transport of the photexcited electron to the Pt cocatalyst, the site of water reduction. Water splitting under visible light irradiation was achieved by Arakawa's group with Ni-doped InTaO_4 modified with RuO_2 or NiO as the cocatalyst.^{47,62} InTaO_4 forms a Wolframite type structure consisting of zigzag chains of edge-shared TaO_6 octahedra (see Figure 3). The Ni^{2+} ions displace In^{3+} from its sites, leading to a contraction of the lattice. The partially filled Ni d-orbitals reduce the band gap of the material from 2.6 to 2.3 eV and cause absorption at 480 nm in the visible. The best catalyst produced H_2/O_2 from pure water under visible irradiation with QE = 0.66% and without deterioration in the activity. Under similar conditions, nondoped NiO– InTaO_4 was 5 times less active. The water-splitting properties of the alkaline-earth tantalates were studied by various researchers.^{147–149} SrTa_2O_6 consists of edge- and corner-shared TaO_6 octahedra (Figure 3) and has a band gap of 4.5 eV based on its absorption edge at 280 nm.¹⁴⁹ from pure water with QE = 7%.^{147–149} BaTa_2O_6 crystallizes in three modifications (hexagonal, 4.0 eV, shown in Figure 3; tetragonal, 3.8 eV; orthorhombic, 4.1 eV), of which the orthorhombic phase shows the highest photocatalytic activities for overall water splitting because it has the largest band gap.⁴⁸ The addition of NiO cocatalysts and of a small amount of $\text{Ba}(\text{OH})_2$ strongly enhanced the photocatalytic reaction. NiTa_2O_6 (network of corner-shared TaO_6 octahedra; Figure 3) also produced both H_2 and O_2 without cocatalysts, whereas MnTa_2O_6 , CoTa_2O_6 , CrTaO_4 , and FeTaO_4 made only traces of H_2 .⁴⁸ $\text{Ca}_2\text{Ta}_2\text{O}_7$ has a Pyrochlore structure consisting of a 3D network of corner-shared TaO_6 octahedra (Figure 3). It showed very good activity for overall water splitting when loaded with a

NiO cocatalyst. The phases Bi_2MTaO_7 ($\text{M} = \text{La}, \text{Y}$) also crystallize in the Pyrochlore structure type, with the lanthanide ions partially occupying Ta^{5+} sites. Based on diffuse-reflectance spectra, the band gaps of these semiconductors are 2.17 eV for $\text{M} = \text{La}$ and 2.22 eV for $\text{M} = \text{Y}$. Under UV irradiation, $\text{Bi}_2\text{LaTaO}_7$ was the slightly more active catalyst, producing H_2/O_2 from pure water.¹⁵⁰ As a Ruddlesden–Popper layered Perovskite with $m = 4$, $\text{Sr}_7\text{Ta}_2\text{O}_7$ is a fairly active catalyst for the overall splitting of water with NiO as the cocatalyst.^{149,151} Because of its larger band gap (4.6–4.8 eV), it is more active than the related $\text{Sr}_2\text{Nb}_2\text{O}_7$. The compounds $\text{M}_2\text{La}_{2/3}\text{Ta}_2\text{O}_7$ ($\text{M} = \text{H}, \text{K}$) and $\text{H}_2\text{SrTa}_2\text{O}_7$ are Ruddlesden–Popper layered Perovskites $m = 2$ (Figure 2) separated by layers of K^+ or H^+ ions. The band gaps of these materials are 3.9–4.0 eV, based on the optical absorption edge around 320 nm. Upon UV irradiation, the NiO-modified catalysts split water into H_2/O_2 . The activity increases in the order $\text{H}_2\text{SrTa}_2\text{O}_7 < \text{K}_2\text{La}_{2/3}\text{Ta}_2\text{O}_7 < \text{H}_2\text{La}_{2/3}\text{Ta}_2\text{O}_7$, with the high rates for NiO– $\text{H}_2\text{La}_{2/3}\text{Ta}_2\text{O}_7$ attributed to efficient NiO and Ni^{2+} intercalation into the interlayer space. Without NiO, the rates are about half, except for $\text{H}_2\text{SrTa}_2\text{O}_7$, which is more active in the pure form.¹⁴⁴ $\text{K}_2\text{Sr}_{1.5}\text{Ta}_3\text{O}_{10}$ belongs to the layered Ruddlesden–Popper phases with $m = 3$ (Figure 2). Its absorption edge occurs at 300 nm, and its band gap is estimated as 4.1 eV. At 77 K, it exhibits a luminescence peak at 404 nm. After modification with RuO_2 , the catalyst has modest activity toward overall water splitting (QE = 2%) under UV light. Without a cocatalyst, the activity is ~ 6 times lower.¹⁵² Under analogous conditions, the 100-layered Perovskite $\text{KBa}_2\text{Ta}_3\text{O}_{10}$ (Figure 2) achieves a QE of 8% after modification with NiO.¹²² High surface modifications of $\text{Sr}_4\text{Ta}_2\text{O}_9$ (cubic Perovskite with Sr on 50% of the Ta sites, not shown) and of the (111)-layered Perovskite $\text{M}_5\text{Ta}_4\text{O}_{15}$ ($\text{M} = \text{Sr}, \text{Ba}$) (Figure 2) were synthesized by calcination of a mixture of the metal ions, citrate, ethylene glycol, and methanol.¹⁴⁹ Their band gaps are 4.8 eV from the respective absorption edges (260 nm). All NiO-modified materials catalyze H_2/O_2 evolution from pure water, with $\text{Ba}_5\text{Ta}_4\text{O}_{15}$ (3.9 eV) having the highest activity when a small amount of $\text{Ba}_{0.5}\text{TaO}_3$ is present.¹⁵³ $\text{K}_3\text{Ta}_3\text{Si}_2\text{O}_{13}$ contains chains of corner-linked TaO_6 octahedra, cross-linked by Si_2O_7 ditetrahedral units (Figure 3). A luminescence at 500 nm (77 K) is probably due to a defect emission, and the band gap is estimated as 4.1 eV based on UV/vis. Under UV illumination, the compound produces small amounts of both H_2 and O_2 , but NiO loading improves the activity by a factor of 10.¹⁵⁴ The crystal structure of $\text{K}_3\text{Ta}_3\text{B}_2\text{O}_{12}$ is very similar, except that the Si_2O_7 units are replaced by double sets of planar BO_3 units (structure not shown). The larger band gap of 4.2 eV (from UV/vis, luminescence at 432 nm) is attributed to distortions of the TaO_6 octahedra with *trans*-O–Ta–O bond angles of 171.5° , compared to 173° in $\text{K}_3\text{Ta}_3\text{Si}_2\text{O}_{13}$. Under UV irradiation, the catalyst is very active toward water splitting but addition of NiO does not improve the activity.¹⁵⁵ In 2004, Arakawa's group studied the effect of the crystal structure on the catalytic activity for a series of tantalates M_3TaO_7 with $\text{M} = \text{Y}, \text{Yb}, \text{Gd}, \text{La}$.¹⁵⁶ For La_3TaO_7 (4.6 eV), it was found that the orthorhombic Weberite modification had a much higher activity than the

cubic Pyrochlore structure, despite the same band gap.¹²³ The Weberite structure type contains chains formed by corner-shared TaO₆ octahedra (Figure 3), whereas the Pyrochlore structure has a 3D network of corner-shared TaO₆ octahedra (Figure 3). For R = Y, Yb (structure contains alternating TaO₈ and LnO₈ cubes; Figure 3) and Gd (Pyrochlore structure, Gd on every second Ta site, not shown), only trace amounts of H₂ were observed. These latter three structures lack TiO₆ chains, which seem necessary for efficient charge transport to the surface of the materials and photocatalytic activity. The tantalates LnTaO₄ (Ln = La, Ce, Pr, Nd, Sm) crystallize in the monoclinic LaTaO₄ type for Ln = La, Ce, Pr, or the monoclinic LaNbO₄ type (Fergusonite) for Ln = Nd, Sm. The structure of the former consists of corner-shared TaO₆ octahedra forming zigzag strings (Figure 3), while the Fergusonite (BiVO₄) structure is a distorted Scheelite (CaWO₄) structure, consisting of TaO₄ tetrahedra separated by La³⁺ ions (Figure 3). The band gaps depend strongly on the nature of the lanthanide ion and vary from 2.4 (Cd) to 3.9 (Nd). Under UV irradiation, only NiO-modified LaTaO₄ has significant catalytic activity for H₂/O₂ evolution from water. The low activities of the other catalysts (they evolve traces of H₂) are explained with trapping of electrons in empty f levels just below the conduction band and with the absence of efficient charge-transport pathways in the case of the Fergusonite structure type.¹⁵⁷ The compounds RbLnTa₂O₇ (Ln = La, Pr, Nd, Sm) are (100)-layered Perovskites of the Dion–Jacobsen series (Figure 2) with *m* = 2 and with band gaps of 3.8–3.9 eV. In addition to the absorption edge at 300 nm, the optical spectra contain sharp absorptions above 300 nm, due to f–f transitions. Under UV irradiation, these catalysts have low activity for the total water splitting, but they improve upon the addition of a NiO cocatalyst. Exchange of Rb⁺ with Na⁺ reduces the rate significantly.^{158,159}

Niobium Oxide and Niobates

While Nb₂O₅ is not an active photocatalyst for water splitting,¹⁶⁰ many niobates do split water upon UV irradiation. One of the best studied catalysts, K₄Nb₆O₁₇, was first discovered by the Domen laboratory in 1986.¹⁶⁰ The layered compound consists of Nb₆O₁₇ sheets (Figure 2) held together by K ions. The sheets are composed of edge- and corner-shared NbO₆ octahedra and the material has a band gap of 3.3 eV. By itself, K₄Nb₆O₁₇ catalyzes nonstoichiometric water splitting at a low rate. Stoichiometric H₂/O₂ evolution is possible after modification with 0.1 wt % NiO, with efficiencies of 5.3–20% under optimized conditions,^{122,161–163} e.g., by the addition of KOH/NaOH to the Ni(NO₃)₂ solution prior to calcination, leading to larger NiO clusters.¹⁶² In comparison, the QE of NiO–Rb₄Nb₆O₁₇ is 10%.⁴³ When K₄Nb₆O₁₇ was internally platinized by ion exchange with [Pt(NH₃)₄]²⁺ followed by reduction, the catalyst produced H₂/O₂ with QE = 1.3%. If Pt was *not* removed from the crystal surface, the activity was ~10 times lower because it catalyzed the back-reaction of H₂ and O₂.¹⁶⁴ Treatment of K₄Nb₆O₁₇ with propylammonium hydrochloride followed by reaction with a soluble titanium(4+) acetate gave a K₄Nb₆O₁₇–TiO₂ composite with low activity for an overall

water splitting.¹⁶⁵ The excited-state dynamics for K₄Nb₆O₁₇ were investigated with transient absorption spectroscopy. Excitation with femtosecond pulses at 266 nm produces a broad transient absorption at 450–700 nm, belonging to photochemically generated charge carriers. The charge carriers recombine following second-order kinetics over the course of 1 ns.¹⁶⁶ Partial exchange of K⁺ in K₄Nb₆O₁₇ with Ni²⁺ or H⁺ does not affect the transient absorption spectra, suggesting that interfacial NbO groups are not involved in the excitation process.¹⁶⁷ Both K₄Nb₆O₁₇ and the related layered oxides KNb₃O₈, KTiNbO₅, and CsTi₂NbO₇ (KNb₃O₈ shown in Figure 2) also catalyze H₂ evolution from an aqueous HI solution. After internal platinization, cation–proton exchange, and sensitization with [Ru(bpy)₃]²⁺ complexes (bpy = 2,2′-bipyridyl derivative), they produce small amounts of H₂ under visible irradiation, with K₂H₂Nb₆O₁₇ being the most active catalyst (QE = 0.3%).^{74,168} The low efficiency in these systems is due to competition of the formed I₃[−] for electrons. The use of (100)-layered Perovskites of the Dion–Jacobsen series A′[A_{*n*−1}B_{*n*}O_{3*n*+1}] (A = K, Rb, Cs; B = Ca, Sr, Na, Pb; *m* = 2–4) as a water-splitting catalyst was first reported in 1990 by Domen's group.¹⁶⁹ These materials consist of double or triple layers of corner-shared NbO₆ octahedra with A ions located in the metal oxide layers and A′ ions between them (Figure 3). From absorption spectroscopy, the band gaps of these materials are estimated as 3.3–3.5 eV.^{38,43} Under UV irradiation and using methanol as a sacrificial donor, the Pt-modified catalysts produce only small quantities of H₂ and no O₂,¹⁶⁹ but the catalytic activity increases dramatically after exchange of the alkaline-metal ions with protons. HCa₂Nb₃O₁₀ produces 5.9 mmol/h/g, and after modification with Pt, the rate increases to 19 mmol/h/g. The effect of cation exchange on the increase in the catalytic activity is attributed to the widening of the interlayer gap to 0.08 nm and the incorporation of water into the interlayer space. Long-chain aliphatic alcohols like ethanol or propanol cannot enter this space and consequently diminish H₂ production.¹⁶⁹ Very recently, Ebina and co-workers reported that, under UV irradiation, KCa₂Nb₃O₁₀ had some activity for H₂ production from pure water (no O₂ was evolved) and medium activity for total water splitting after intercalation of RuO_x between the layers.¹⁷⁰ Treatment of HCa₂Nb₃O₁₀ with alkylamines followed by treatment with tetraethoxysilane produced a HCa₂Nb₃O₁₀–SiO₂ composite with good activity for H₂ evolution from aqueous methanol. Added Pt further increased the activity. Because of the increased interlayer space, longer-chain alcohols could serve as efficient electron donors.¹⁷¹ The activity of KCa₂Nb₃O₁₀ could also be enhanced significantly by restacking exfoliated HCa₂Nb₃O₁₀ nanosheets with KOH or NaOH, producing a material with 10 times greater specific surface area.¹⁷² When Pt particles were intercalated, the restacked catalyst evolved H₂ 10 times faster from an aqueous methanol solution than Pt-surface-modified KCa₂Nb₃O₁₀. When RuO₂ was intercalated, water was split stoichiometrically at a medium rate.¹⁷⁰ The water-splitting activity and the excited-state dynamics of exfoliated HCa₂Nb₃O₁₀ nanosheets were studied by Osterloh and co-workers.¹⁷³ After excitation at 300 nm, electrons and holes recombine over the course of nanoseconds

following second-order kinetics. Under UV irradiation, the nanosheets evolve only low H_2 rates from water ($QE = 0.22\%$), but the growth of Pt nanoparticles on the nanosheets enhances the catalytic activity more than 30 times ($QE = 7.5\%$). Doping of the Dion–Jacobsen phase $K_{0.5}La_{0.5}Ca_{1.5}Nb_3O_{10}$ (3.44 eV) with PbO/Bi_2O_3 gives $K_{0.5}La_{0.25}Bi_{0.25}Ca_{0.75}Pb_{0.75}Nb_3O_{10}$ with a band gap of 3.06 eV. After modification with Pt nanoparticles, the catalyst produces traces of H_2 from aqueous methanol and medium rates of O_2 from aqueous $AgNO_3$ under visible light irradiation.¹²⁷ The smaller band gap (2.5 eV) of the isomorphous $RbPb_2Nb_3O_{10}$ allows visible light absorption at <500 nm. After modification with Pt, only traces of H_2 are evolved from aqueous methanol, but protonation and internal platinization enable good H_2 rates.¹⁷⁴ $PbBi_2Nb_2O_9$ is a member of the Aurivillius-type layered Perovskites with $m = 2$ (Figure 2). Its band gap is 2.88 eV, judged from the visible absorption edge at 431 nm. Under visible light, the Pt-loaded catalyst produces small rates of H_2 ($QE = 0.95\%$) from aqueous methanol and large rates of O_2 ($QE = 29\%$) from aqueous $AgNO_3$.¹¹³ For the isostructural Bi_3TiNbO_9 , higher H_2 but lower O_2 rates are observed under UV irradiation.¹⁷⁵ Similar to the tantalates, the niobates Bi_2MNbO_7 ($M = Al, Ga, In$) crystallize in the Pyrochlore structure type (Figure 3) and have band gaps of ~ 2.7 – 2.9 eV.^{47,176} Under UV irradiation, they evolve H_2 from aqueous methanol and O_2 from a $Ce(SO_4)_2$ solution at rates up to 710 and 25 $\mu\text{mol/h/g}$, respectively. The catalyst with $M = Al$ is the most active. In contrast, the compounds M_2BiNbO_7 ($M = Ga, In$, Pyrochlore structure) have the ability to split water stoichiometrically under UV irradiation. The enhanced reactivity is observed despite the fact that the band gaps of these compounds are smaller (2.5–2.6 eV) than those for the series Bi_2MNbO_7 .¹⁷⁷ $ZnNb_2O_6$ crystallizes in the columbite structure type containing double layers of edge- and corner-shared NbO_6 octahedra (Figure 3).¹⁷⁸ Under band-gap (4.0 eV) irradiation, it produces small amounts of H_2 but no O_2 . Using a NiO cocatalyst, near-stoichiometric water splitting at a low rate can be achieved.¹⁷⁸ $SnNb_2O_6$ (Figure 4) forms the layered Fordite structure consisting of double layers of edge-shared NbO_6 octahedra separated by layers of Sn^{2+} ions. It has an absorption edge at 540 nm and a band gap of 2.3 eV. After modification with Pt, it evolves small amounts of H_2 from an aqueous methanol solution under visible light.¹⁷⁹ The isostructural $SnTa_2O_6$ is not active under these conditions but evolves small amounts of H_2 under UV irradiation. $Sr_2Nb_2O_7$ (3.9 eV from UV/vis) belongs to the Ruddlesden–Popper layered Perovskites with $m = 4$ (Figure 2). With NiO, the material splits water into H_2/O_2 with medium activity, while without NiO, only H_2 is produced.¹⁵¹ QEs of up to 23% were observed by Lee and co-workers.¹²² The isostructural $Ca_2Nb_2O_7$ (4.3 eV) produced H_2 with $QE = 7\%$ after modification with NiO (the O_2 rate was not determined).¹²² $Cs_2Nb_4O_{11}$ has a complex structure containing corner- and edge-shared NbO_6 octahedra and NbO_4 tetrahedra (Figure 3). Despite its large band gap of 3.7 eV, it emits a blue photoluminescence at 440 nm at 77 K. Depending on the temperature for NiO coating, the catalyst

can reach activities as high as $K_4Nb_6O_{17}$.¹⁸⁰ $Ba_5Nb_4O_{15}$ is a (111)-layered Perovskite containing layers of four NiO_6 octahedra thick slabs separated by Ba^{2+} cations (Figure 2). Its absorption edge lies at 322 nm (the band gap is 3.9 eV), and at 77 K it produces a broad photoluminescence at 532 nm. Its excellent activity for water splitting under UV light irradiation could be optimized further by increasing the surface area of the catalyst using a modified synthesis.¹⁸¹ The isostructural tantalates are also excellent catalysts for overall water splitting.^{149,153}

Other Transition Metal Oxides

Other transition metal oxides catalyze partial water redox reactions with the aid of suitable electron donors or acceptors. Fergusonite, $BiVO_4$, consists of VO_4 tetrahedra separated by Bi^{3+} cations (Figure 3). With a 2.3 eV band gap, it shows some activity for O_2 evolution from aqueous solutions of $AgNO_3$ under visible light irradiation.¹⁸² Ag_3VO_4 also contains MO_4 tetrahedral units (structure not shown) and has a band gap of 2.0 eV and an absorption edge at 570 nm. Under visible light illumination, it evolves O_2 from an aqueous $AgNO_3$ solution at good rates.¹⁸³ WO_3 (2.8 eV) crystallizes in the ReO_3 structure type (not shown). In the presence of Fe^{3+} or Ag^+ ions, it can oxidize water at fairly high rates but not reduce it. In the presence of methanol, WO_3 bronze forms instead of H_2 .^{184,185} Pt-modified WO_3 was also used by Abe et al. in a tandem system with Cr/Ta-doped Pt- $SrTiO_3$ for an overall water splitting.^{118,186} Under visible light irradiation, Pt- WO_3 alone with $NaIO_3$ produced O_2 at a high rate but no H_2 . Combined with Pt/ $SrTiO_3$ and using iodide as a redox shuttle, water splitting was observed under monochromatic light (420.7 nm) with small H_2/O_2 rates of 0.21 and 0.11 $\mu\text{mol/h}$ for 0.4 g of both catalysts ($QE = 0.1\%$). When Pt/ $SrTiO_3$ was replaced with TaON (see below), QE increased to 0.4%.¹⁸⁷ The tungstate $Na_2W_4O_{13}$ forms a layered solid composed of two WO_6 octahedra thick sheets separated by Na ions (Figure 2). Under UV light, the material has good activity for H_2 evolution from aqueous MeOH and for O_2 evolution from aqueous $AgNO_3$.¹⁸⁴ In comparison, the Keggin ion $[Si(W_3O_{10})_4]^{4-}$ (Figure 4), a homogeneous catalyst, produces high rates of H_2 from aqueous methanol but no O_2 even when $AgNO_3$ is present.¹⁸⁴ Apparently, the clusters lack active sites for O_2 evolution. Ca_2NiWO_6 (structure not shown) crystallizes in the Perovskite structure, with Ni^{2+} and W^{6+} ions occupying positions in alternating (011) layers. Based on optical data, it has a band gap of 2.60 eV. Under UV irradiation, the Pt-modified semiconductor evolves small amounts of H_2 from aqueous methanol and very small amounts of O_2 from aqueous $AgNO_3$.¹⁸⁸ Bi_2WO_6 belongs to the $(Bi_2O_2)^{2+}[A_{m-1}B_mO_{3m+1}]^{2-}$ Aurivillius structure type with alternating layers of corner-shared WO_6 and Bi_2O_2 layers (see Figure 2). $Bi_2W_2O_9$ is a defect structure, with the A sites remaining empty. After modification with Pt, it is slightly more active in an aqueous methanol solution than Bi_2WO_6 (UV irradiation). Under visible light, Bi_2WO_6 (2.8 eV) also catalyzes slow O_2 evolution from an aqueous $AgNO_3$ solution.¹⁷⁵ While the low-temperature (LT) modification of Bi_2MoO_6 (2.7 eV) is isomorphous with Bi_2WO_6 from above, the HT

form (3.02 eV) contains isolated Mo_4 tetrahedra (Figure 4).¹⁸⁹ Under UV and visible irradiation, both catalyze O_2 evolution from aqueous AgNO_3 , with the LT form having higher activity than the HT form, presumably because of the lack of an efficient charge-transport pathway in the latter. No H_2 evolution from aqueous MeOH is observed for either material.¹⁸⁹ The structures (not shown) of $\text{Bi}_2\text{Mo}_2\text{O}_9$ (3.10 eV) and $\text{Bi}_2\text{Mo}_3\text{O}_{12}$ (2.88 eV) are similar to the HT form, and their activities for O_2 evolution from aqueous AgNO_3 are intermediate between the above materials.¹⁸⁹ The only other molybdates known to split water are PbMoO_4 and Cr/PbMoO_4 .^{190,191} Both crystallize in the Scheelite (CaWO_4) structure type, which contains WO_4 tetrahedra isolated by the Ca ions (Figure 4). Both have band gaps of 3.31 eV (PbMoO_4) and 2.16–2.43 eV (Cr/PbMoO_4), depending on the Cr content. Pt/PbMoO_4 , but not any of the Cr-doped phases, evolves good rates of H_2 from an aqueous methanol solution under UV light. From pure water, O_2 evolution is fair, but no H_2 is formed. From aqueous AgNO_3 , both the Cr-doped and pure phases evolve good rates of O_2 ,¹⁹⁰ with the performance of the Cr phases extended into the visible (>420 nm).¹⁹¹ The band gap of CeO_2 (Fluorite structure, not shown) is between 2.79 and 3.18 eV, depending on the preparation. Under UV irradiation, CeO_2 catalyzes slow O_2 evolution from an aqueous $\text{Fe}_2(\text{SO}_4)_3$ solution.¹⁸⁵ When prepared by hydrolysis of CuCl , cuprous oxide Cu_2O (cuprite; ccp O lattice with Cu in $\frac{1}{4}$ of the T_d voids; structure not shown) evolves small but stoichiometric rates of H_2/O_2 under visible light irradiation from pure water. Cu_2O has a band gap of 2.0 eV (from UV/vis spectroscopy).¹⁹² Later experiments showed that H_2/O_2 evolution is at least partially due to *mechanocatalysis*. This presently not understood process is driven by frictional energy released into the reaction mixture by stirring it with a stir bar.¹⁹³ Several binary metal oxides (Cu_2O , NiO , CO_3O_4 , and Fe_3O_4) showed small activity for water splitting under mechanocatalytic conditions, while much lower, yet definite activity was also observed for RuO_2 and IrO_2 . Cupric oxide (CuO) and Fe_2O_3 did not show activity for the reaction, and neither did TiO_2 , ZnO , and WO_3 .

Main-Group Metal Oxides

In very recent work, Inoue and co-workers demonstrated that the oxides of the main-group elements Ga, In, Ge, Sn, and Sb are effective photochemical water-splitting catalysts, although often only in the presence of suitable redox agents. In_2O_3 (2.7 eV) crystallizes in the $\text{C-M}_2\text{O}_3$ structure type (a defect Fluorite structure with $\frac{1}{4}$ of O removed, not shown).⁶⁶ After modification with a Pt cocatalyst, the material evolves small quantities of H_2/O_2 from aqueous solutions of methanol/ AgNO_3 , respectively, under visible light irradiation.¹⁹⁴ With NiO as a cocatalyst, small quantities of H_2 are formed from pure water (UV light). $\text{Ba}_2\text{In}_2\text{O}_5$ crystallizes in the Brownmillerite structure type consisting of alternating layers of corner-shared InO_4 tetrahedra and corner-shared InO_6 octahedra and Ba^{2+} ions between these units (Figure 4). Under visible irradiation, Pt-modified $\text{Ba}_2\text{In}_2\text{O}_5$ evolves small amounts of H_2/O_2 from aqueous solutions of methanol and AgNO_3 , while NiO -modified $\text{Ba}_2\text{In}_2\text{O}_5$ evolves H_2 from pure water. Doping with Cr ions reduces the rates slightly,

however, if the Cr-doped material is mixed with $\text{Cr/In}_2\text{O}_3$, the activity for H_2/O_2 evolution under visible light or UV is enhanced.¹⁹⁴ $\text{In}_2\text{O}_3(\text{ZnO})_m$ ($m = 3$ and 9) crystallizes in a layered structure containing sheets of octahedral InO_6 units alternating with sheets of tetrahedral and trigonal-pyramidal ZnO_4 units (Figure 4).¹⁹⁵ The band gaps are 2.6/2.7 eV for $m = 3$ and 9 , respectively. Under visible light irradiation, both materials evolve small quantities of H_2/O_2 from aqueous solutions of methanol and AgNO_3 .¹⁹⁶ CaIn_2O_4 has a tetragonal crystal structure containing a network of edge-shared distorted InO_6 octahedra (Figure 4). Under UV irradiation, the RuO_2 -modified semiconductor has some activity for overall water splitting, depending on the calcination temperature and RuO_2 loading. By comparison, the activities of RuO_2 -modified SrIn_2O_4 (orthorhombic; Figure 4) and LaInO_3 are by factors of 3 and 10 lower.¹⁹⁷ While separate Ga_2O_3 and Lu_2O_3 evolve only traces of H_2 from aqueous methanol, the mixture of the two oxides forms an active ($\text{QE} = 6.81\%$ at 320 nm) water-splitting catalyst, after doping with Zn and modification with NiO particles.¹⁹⁸ The structure of $\beta\text{-Ga}_2\text{O}_3$ (Figure 4) consists of corner- and edge-shared GaO_6 octahedra and GaO_4 tetrahedra. Its absorption edge is 270 nm, suggesting a band gap of 4.6 eV. Lu_2O_3 crystallizes in the $\text{C-M}_2\text{O}_3$ structure type similar to In_2O_3 (see above). ZnGa_2O_4 is also known to catalyze the evolution of H_2 and O_2 from pure water under UV irradiation. After modification with RuO_2 , H_2 and O_2 are evolved at near-stoichiometric but low rates. The catalyst activity strongly depends on the calcination temperature. The crystal structure of ZnGa_2O_4 (Figure 4) contains GaO_6 octahedra and ZnO_4 tetrahedra. The compound has a band gap of 4.3 eV from UV/vis spectra (2.78 eV from DFT calculations).¹⁹⁹ Sr_2SnO_4 can be viewed as a Ruddlesden–Popper phase with $m = 1$ (Figure 2). It is the only tin oxide known to split water photochemically, albeit at low rates, under UV irradiation, and only after modification with RuO_2 .²⁰⁰ For Sb, several oxides have been identified as water-splitting catalysts. After the addition of a RuO_2 cocatalyst, $\text{M}_2\text{Sb}_2\text{O}_7$ ($\text{M} = \text{Ca}$, Sr , Weberite), CaSb_2O_6 , and NaSbO_3 (Ilmenite) all catalyze overall water splitting with near-stoichiometric O_2 evolution.^{200,201} NaSbO_3 crystallizes in the Ilmenite structure containing layers of SbO_6 octahedra separated by Na ions (Figure 2). $\text{M}_2\text{Sb}_2\text{O}_7$ ($\text{M} = \text{Ca}$, Sr), on the other hand, crystallize in the Weberite structure type (Figure 4), containing corner-shared SbO_6 octahedra in two different environments. After modification with RuO_2 , the activities remain generally low but increase in the order $\text{CaSb}_2\text{O}_6 < \text{NaSbO}_3 < \text{Ca}_2\text{Sb}_2\text{O}_7 < \text{Sr}_2\text{Sb}_2\text{O}_7$. This trend is correlated with the presence of increasingly distorted MO_6 octahedra in the more active structures.²⁰¹ Only traces of H_2/O_2 are evolved without a RuO_2 cocatalyst. In contrast to the alkaline-earth germanates (M_2GeO_4 , $\text{M} = \text{Ca}$, Sr , Ba), Zn_2GeO_4 is stable under UV irradiation. Its Willemite structure (Figure 4) consists of a 3D network of corner-shared GeO_4 and ZnO_4 tetrahedra. Based on the absorption edge at 310 nm, the band gap is 4.0 eV. Under UV light, the RuO_2 -modified catalyst evolves H_2 and O_2 in near-stoichiometric quantity from pure water.²⁰²

Metal Nitrides and Phosphides

In recent years, nitrides and nitride/oxide compounds of Ga and Ge have evolved as promising water-splitting catalysts that can operate under visible light and without external redox agents.^{45,46} GaN:ZnO forms a solid solution in the Wurtzite structure type (not shown). Interestingly, the band gap of the solution (2.38 eV) is smaller than that of the separate components, GaN (3.4 eV) and ZnO (3.2 eV), which appears to be due to a raised valence band edge that results from p–d repulsion of Zn 3d and N 2p electrons. After modification with RuO₂, Ga_{0.38}N_{0.33}Zn_{0.13}O_{0.16} splits water stoichiometrically into H₂ and O₂ at very good rates under UV and with reasonable rates under visible light (QE = 0.14% for 300–480 nm).^{203,204} RuO₂ can be replaced with other cocatalysts (Rh, Ir, and Pt),²⁰⁵ among which a mixed Cr/Rh oxide is the most effective (QE = 2.5%).^{14,15} β -Ge₃N₄ crystallizes in the Phenacite structure type (see Figure 3), consisting of corner-shared GeN₄ tetrahedra. Despite its large band gap of 3.8–3.9 eV (absorption edge at 300 nm), the compound gives a broad luminescence at 480 nm (at 77 K). After modification with 20–50 nm RuO₂ particles, the semiconductor splits pure water stoichiometrically into H₂/O₂ with excellent activity under UV light (QE = 9%), while pure Ge₃N₄ is catalytically inactive.^{206,207} Various Ga/Zn oxynitrides with band gaps of 2.66–4.31 eV can be obtained by doping ZnGeN₂ (wurtzite) with ZnO. Of these, the solid solution Zn_{1.44}GeN_{2.08}O_{0.38} has a band gap of 2.7 eV. Upon modification with 1–5 wt % RuO₂ but not without it, the material catalyzes stoichiometric water splitting at medium rates. Under visible light, the activity is about 4 times lower.²⁰⁸ Neither ZnO nor ZnGeN₂ split water under similar conditions. Ta₃N₅ crystallizes in the pseudo-Brookite structure (Figure 3) with edge- and corner-shared irregular TaN₆ octahedra. The O-doped form Ta₃N_{4.8}O_{0.3} with a 2.1 eV band gap and a 600 nm absorption edge is obtained by the reaction of Ta₂O₅ with NH₃. After modification with Pt, this catalyst evolves very small rates of H₂ from aqueous MeOH under visible irradiation (QE = 0.1% at 420–600 nm). Prior calcination with La₂O₃ (buffers at pH = 8.5) leads to good rates of O₂ from aqueous AgNO₃, while without La₂O₃, N₂ is formed as a result of anodic dissolution of the catalyst under acidic conditions.^{69,209} β -TaON is isomorphous with Baddeleyite, monoclinic ZrO₂ (Figure 4) and consists of edge-shared TaO₇ polyhedra. The reaction of Ta₂O₅ with NH₃ at 1123 K for 15 h affords a material with the composition TaO_{1.24}N_{0.84},²¹⁰ whose band gap of 2.5 eV is estimated from a 500 nm absorption edge. Under visible light, the catalyst is quite active for O₂ evolution (QE = 34%) from aqueous AgNO₃, when La₂O₃ is added as a base buffer (pH = 8). With Pt, Rh, or Ir cocatalysts, the material evolves only small rates of H₂ from aqueous methanol (QE = 0.2%),²¹⁰ but with Ru, the activity increases significantly (QE = 0.8%).^{44,211} Abe et al. found that, under visible light, TaON evolves small amounts of H₂ from an aqueous iodide solution.¹⁸⁷ In tandem with Pt–WO₃, stoichiometric water splitting is possible in the presence of iodide as a redox shuttle. The QE = 0.4% at 420 nm remained stable for up to 100 h.¹⁸⁷ The oxynitrides MTaO₂N (M = Ca, Sr, Ba) all crystallize in the Perovskite structure type. The band gaps decrease with increasing the

radius of the alkaline-earth metal: 2.5, 2.1, and 2.0 eV for M = Ca, Sr, and Ba, respectively. Under visible light, H₂ was formed from aqueous methanol, but no O₂ evolution took place, not even from aqueous AgNO₃.⁴⁴ LaTiO₂N also has the Perovskite structure and a band gap of 2.1 eV. Under visible light, the Pt-modified catalyst slowly evolved H₂ from aqueous methanol (QE = 0.15%), and in the presence of AgNO₃ and La₂O₃ as a base buffer (pH = 8), O₂ is produced at reasonable rates during the first 10 h. The formation of N₂ can be suppressed by doping of the catalyst with La and by the addition of 2 wt % colloidal IrO₂ as a cocatalyst. Under optimized conditions, this catalyst evolves O₂ with QE = 5%.²¹² Y₂Ta₂O₅N₂ crystallizes in the Pyrochlore structure type (Figure 3). The prepared Y₂Ta₂O_{4.94}N_{2.18} is slightly nonstoichiometric and has an absorption edge of 560 nm, suggesting a band gap of 2.2 eV (compared to 3.8 nm for YTaO₄). Using visible light, the Pt/Ru-modified compound evolved good rates of H₂ from aqueous ethanol. The material containing separate Ru or Pt cocatalysts gave lower activity. Good O₂ evolution also occurred from aqueous AgNO₃ when La₂O₃ was added as a buffer. Only a small amount of N₂ was detected at the beginning of the reaction.²¹³ InP crystallizes in the zinc blende structure type (not shown) and has a band gap of 1.25 eV. With Pt as a cocatalyst, it evolves small quantities of H₂ from aqueous solutions containing sulfite and sulfide as electron donors when illuminated with UV light.²¹⁴ The catalytic nature of the process is doubtful.

Metal Sulfides

Metal sulfides are attractive as photochemical water-splitting catalysts because of their small band gaps that allow absorption of visible light. However, applications as catalysts have been hampered by the photochemical instability requiring the use of sacrificial donors for photochemical H₂ evolution from water. CdS (Wurtzite structure, not shown) is probably the best studied metal sulfide photocatalyst.^{27,29,32,33} Because of its relatively narrow band gap (2.4 eV), it absorbs visible light at wavelengths of <510 nm. The flat-band potential of CdS (–0.87 V)⁷⁰ is sufficiently high to reduce H₂O, and the top of its valence band (1.5 V vs NHE) is theoretically suitable to allow oxidation of water. For 4 nm CdS nanoparticles, the lifetime of photogenerated charge carriers is on the order of 50 ps.²¹⁵ However, prolonged irradiation of CdS suspensions leads to photocorrosion of CdS into Cd²⁺ and S (sulfate in the presence of O₂).^{71,216,217} This reaction can be suppressed by the addition of reducing agents to the aqueous phase (see below). Darwent and Porter^{218,219} and separately Mills and Porter⁹³ were the first to investigate CdS for photochemical water splitting using EDTA as the sacrificial agent. Under visible light irradiation, Pt–CdS powder evolved H₂ from aqueous EDTA with QE = 4%, but without Pt, the activity was reduced by a factor of 10. Prolonged irradiation (>4 h) leads to decomposition of the catalyst.^{93,219} In 1984, Reber and co-workers published the most comprehensive study on photocatalysis with CdS–Pt microcrystalline powders.²²⁰ Among other variables, they studied the effect of various sacrificial electron donors and their concentrations, reaction temperature, pH, and of

irradiation wavelengths. The best catalysts evolved H_2 at 2.9 mmol/h/0.4 g (QE = 25%) under irradiation with light of >300 nm in the presence of either S^{2-} , SO_3^{2-} , or $\text{S}^{2-}/\text{HPO}_4^{2-}$ as reducing agents. The activity dropped by 21% over the course of 6 days, likely because of deactivation of Pt due to formation of Pt-S species. Under solar irradiation, 50 mL/h of H_2 were produced with 0.4 g of catalyst suspension equal to an energy efficiency of 2%. In a follow-up study, the same team showed that the activity of CdS could be increased up to 357 mL/h/0.4 g (QE = 37%) of H_2 in aqueous $\text{Na}_2\text{S}/\text{Na}_2\text{SO}_3$ by doping CdS with 15 mol % ZnS and by the addition of a Pt cocatalyst.²¹⁷ The effect of Ag_2S coprecipitates, the specific surface area of the catalyst, and material decomposition were also tested. At about the same time, Gratzel's group published results on Pt–CdS– RuO_2 colloidal catalysts operating under visible light.²²¹ The CdS particles were synthesized in situ and identified via their absorption edge at 520 nm. Cofunctionalization with RuO_2 and Pt gave a catalyst that produced stoichiometric amounts of H_2/O_2 from pure water under visible light irradiation. The H_2 evolution increased with temperature. Without RuO_2 , CdS was found to decompose rapidly with the formation of S. Similar results were obtained with CdS–Rh– RuO_2 and CdS–Pt– RuO_2 catalysts in a different laboratory.²²² In general, O_2 evolution from CdS-based catalysts is problematic.^{21,70} To solve this problem and to address the photochemical instability of CdS, more recent studies have focused on other forms of CdS, including CdS nanoparticles stabilized in micelles,^{223–225} CdS composites with other semiconductors (TiO_2 ,^{226,227} ZnS,^{228,229} and CdSe²³⁰), metal cocatalysts (Pt, Pd, Rh, Ru, Ir, Fe, Ni, and Co),^{231,232} hollow CdS microparticles,²³³ and Cu-doped CdS.^{234,235} The effect of CdS preparation was also studied.^{236,237} Despite these efforts, photocorrosion and the inability of water oxidation remain the principle problems of CdS-based water-splitting catalysts.

ZnS (zinc blende) is the other major metal sulfide investigated for photochemical water splitting. It has a band gap of 3.66 eV, which restricts light absorption to the UV (<340 nm). Similar to CdS, it undergoes photochemical decomposition into the components when irradiated in the absence of sacrificial electron donors.¹³ The first report on water splitting with ZnS was published by Yamagida in 1983,²³⁸ who synthesized ZnS from $\text{Zn}(\text{SO}_4)$ or ZnCl_2 and Na_2S in water. With tetrahydrofuran (THF) or alcohols as the sacrificial donor, this catalyst produced H_2 gas under UV irradiation. The addition of D_2O caused the production of D_2/H_2 with a ratio of 7, establishing water as the H_2 source. The most comprehensive study on photocatalysis of ZnS and ZnS–Pt was carried out by Reber et al.¹³ The effect of catalyst preparation, sacrificial electron donors, pH, and temperature was investigated. Under irradiation with >300 nm light and at 60°C , ZnS–Pt catalyzed H_2 evolution with quantum yields of up to 90% from aqueous solutions of sulfide and sulfite. Importantly, it was found that, in the presence of SO_3^{2-} , metallic Zn was formed, which was thought to assume a role in electron transfer to water. Long-time catalytic tests showed that no deactivation of ZnS occurred over 34 h. To improve visible light absorption of ZnS, Kudo's group has tested metal dopants (Cu,²³⁹ Ni,²⁴⁰

and Pb^{241}). Doping can move the absorption edge to 500 nm in the case of Ni^{2+} ions and 550 nm in the case of Pb^{2+} ions. Under visible irradiation, a Ni^{2+} -doped catalyst produced H_2 from aqueous $\text{K}_2\text{SO}_3/\text{Na}_2\text{S}$ with QEs of up to 1.3%.²⁴⁰ Cu-doped ZnS gave QE = 3.7% under visible light irradiation from aqueous Na_2SO_3 .²³⁹ Doping of ZnS with variable amounts of AgInS_2 or CuInS_2 produces a series of solid solutions that crystallize in the cubic zinc blende or hexagonal Wurtzite structure.^{242–245} The optical absorption of these materials can be adjusted between 400 and 800 nm, depending on the composition.²⁴³ Visible light irradiation of the Pt- or Ru-derivatized catalysts produces H_2 with a QE of up to 7.5% from aqueous $\text{Na}_2\text{S}/\text{K}_2\text{SO}_3$.²⁴³ For the Pt-loaded $\text{Ag}_{0.22}\text{In}_{0.22}\text{Zn}_{1.56}\text{S}_2$ photocatalyst, a QE of 20% was measured at 420 nm under solar irradiation conditions.²⁴⁵

While pure InS or In_2S_3 do not catalyze photochemical water splitting, several ternary indium sulfides do. $\text{Na}_5\text{In}_7\text{S}_{13}$ forms a Zeolite-like structure containing SIn_4 tetrahedral units (structure not shown) and has a band gap of 3.2 eV.²⁴⁶ Under irradiation with a Xe lamp, the material evolves a fair rate of H_2 from a 0.5 M Na_2SO_3 aqueous solution. Using Na_2S as the sacrificial electron donor, $\text{Na}_{14}\text{In}_{17}\text{Cu}_3\text{S}_{35} \cdot x\text{H}_2\text{O}$ evolves small quantities of H_2 under visible light irradiation, equal to QE = 3.7% at 420 nm. With SO_3^{2-} , the QE drops to 0.37%. For $(\text{AEP})_6\text{In}_{10}\text{S}_{18}$, AEP = protonated 1-(2-aminoethyl)piperazine, higher rates were observed.²⁴⁷ The latter two compounds form complex networks of MS supertetrahedra (example in Figure 4). S and Zn doping of $\text{In}(\text{OH})_3$ (ReO_3 structure, not shown) produces $[\text{In}(\text{OH})_y\text{S}_z]$ and $\text{In}(\text{OH})_y\text{S}_z\text{:Zn}$ ($z \sim 2$) with absorption edges at 570 and 470 nm, respectively. Both catalysts evolve H_2 from aqueous sulfide/sulfite solutions under visible light illumination, with Pt-loaded $\text{Zn:In}(\text{OH})_y\text{S}_z$ reaching QE = 0.59%.²⁴⁸ CuInS_2 (zinc blende structure with Cu^+ and In^{3+} on tetrahedral sites, not shown) and CuIn_5S_8 (spinel structure; ccp lattice of S^{2-} with Cu^+ only in T_d voids and In^{3+} in both O_h and T_d voids) produce very low amounts of H_2 in the presence of sulfite and under UV light irradiation.²⁴⁹ NaInS_2 has a layered structure consisting of layers of edge-shared octahedral InS_6 units (Figure 2). Its 2.3 eV band gap corresponds to an absorption edge of 550 nm. Under visible light irradiation, the Pt-modified catalyst showed good photocatalytic activity for H_2 evolution from an aqueous K_2SO_3 solution.²⁵⁰ WS_2 forms the MoS_2 structure (not shown) containing layers of trigonal-prismatic WS_6 units. It has a 1.7 eV direct band gap and a 1.3 eV indirect band gap. When supported on SiO_2 and using fluoresceine as a sensitizer, the catalyst produced H_2 from aqueous EDTA solutions under visible light.²⁵¹ Bi_2S_3 crystallizes in the Bismuthinite structure type containing chains of corner-shared BiS_4 tetrahedra (Figure 4). It has a band gap of 1.28 eV. Under visible irradiation, it produces H_2 at intermediate rates from an aqueous sulfide solution. Rates decline after 100 min, probably because of disulfide formation. Platinization improves the activity by 25%.²⁵² Other metal sulfides have also been tested.²³⁴ Of these, In_2Se_3 , SnS_2 , HgS , Ti_2S , PdS , EuS , CuS , FeS , CoS , and Fe_2S_3 were found to be inactive because of their small band gaps (<2 eV).

Summary

1. Only metal compounds with d^0 ions (Ti, Zr, Nb, and Ta) and d^{10} ions (Ga, In, Ge, Sn, and Sb) have activity for overall photochemical splitting of water. Oxides are dominant, but nitrides and oxynitrides (GaN:ZnO and Ge_3N_4) are also known to catalyze the reaction.

2. Other catalysts require reducing agents (alcohols, sulfides, sulfites, and EDTA) or oxidizing agents (persulfate and Ce^{4+} or Ag^+ ions) to facilitate either water reduction or oxidation. An overall water splitting might be achievable with these materials by coupling of appropriate semiconductor pairs into tandem systems according to Figure 1D (for examples, see refs 107, 108, 118, and 187).

3. Semiconductors containing metal ions with partially filled orbitals generally show reduced or no catalytic activity, because the ions act as catalytic centers for electron-hole recombination.^{48,124,190,191,194} Exceptions are diamagnetic ions with the d^{10} configuration (Ag^+ , Zn^{2+} , and Cu^+), as seen in various semiconductors (Table 1), and ions with s^2 configuration (Pb^{2+} and Bi^{3+}), as in PbTiO_3 and $\text{PbBi}_4\text{Ti}_4\text{O}_{15}$,¹²⁷ $\text{PbBi}_2\text{Nb}_2\text{O}_9$,¹¹³ PbMoO_4 ,¹⁹⁰ and Pb:ZnS .²⁴¹ The fact that electron-hole recombination seems to be less dominant for Ni^{2+} ions (e.g., in Ni:InO_4 ,^{47,62} NiTa_2O_6 ,⁴⁸ and Ni:ZnS ,²⁴⁰) has been attributed to the difficulty of accessing $1+/3+$ oxidation states for this ion.⁴⁸

4. All metal sulfides, including CdS and ZnS, undergo photochemical decomposition in the absence of sacrificial electron donors.^{13,93,219}

5. The only visible-light-driven catalysts that split water and do not require external redox agents are $\text{NiO/RuO}_2\text{-Ni:InTaO}_4$ (QE = 0.66%),^{47,62} the Pt-WO_3 and $\text{Pt/SrTiO}_3/\text{TaON}$ (QE = 0.1%) tandem system,^{118,186,187} Cr/Rh-GaN:ZnO (QE = 2.5%),^{14,15,205} and $(\text{Zn}_{1.44}\text{Ge})(\text{N}_{2.08}\text{O}_{0.38})$ (QE not determined).²⁰⁸

6. Trends in H_2/O_2 evolution rates roughly follow the size of the semiconductor *band gaps*. Group 13/14 element oxides (2.7–3.8 eV) are generally less active than titanates (TiO_2 , 3.0–3.1 eV). Niobates (Nb_2O_5 , 3.1–3.5 eV) are more active than titanates but less active than tantalates (Ta_2O_5 , 4.0–4.6 eV). Here, larger bandgaps reflect an increasing thermodynamic driving force for water splitting. The low activity of ZrO_2 in spite of its large band gap (5.0–5.7 eV) may be explained with poor spectral overlap of its absorption (<217–248 nm) with commonly used Xe and Hg light sources. The low activity of Sr_2SnO_4 (3.8 eV) is likely a result of the low flat-band potential, which is ineffective for proton reduction.

7. Activities strongly depend on *cocatalysts*. In the majority of cases, cocatalysts enable (e.g., Ta_2O_5)^{40,48} or increase the water-splitting activity of a semiconductor. Many semiconductors evolve H_2 from water but O_2 only in the presence of a cocatalyst (often NiO). This behavior is likely due to the greater complexity of O_2 evolution (a four-electron process) compared to H_2 (a two-electron process). Very few semiconductors experience a decrease in activity upon attachment of a cocatalyst. For example, ZrO_2 is more active without Pt, Cu, Au, or RuO_2 cocatalysts,¹³⁷ as is the Ruddlesden-Popper phase $\text{H}_2\text{SrTa}_2\text{O}_7$.¹⁴⁴ These observations may be explained with the large band gaps of these semiconductors and possible

electron-hole recombination on the metal cocatalyst. In other cases, low activities are caused by back-reaction of H_2 and O_2 (the reverse of reaction 1), a problem particularly relevant for Pt.⁹⁵ Au particles do not promote this back-reaction but catalyze the reduction of O_2 in competition with H_2O .⁹⁷ For many catalysts, the activity is strongly dependent on the cocatalyst material (see, for example, $\text{K}_2\text{La}_2\text{Ti}_3\text{O}_{10}$ ^{132,133} and GaN:ZnO ^{14,15,203–205}) and on the methods of cocatalyst attachment (see SrTiO_3 ,¹¹⁷ $\text{Cs}_2\text{Nb}_4\text{O}_{11}$,¹⁸⁰ and CdS ²²⁰). These dependencies must be attributed to variations in charge transport across the cocatalyst–semiconductor interface.

8. *Additives* (other than sacrificial redox agents) often enhance the activity of the catalyst. Base buffers, like La_2O_3 , can stabilize sensitive metal nitrides against photochemical decomposition.^{44,69,209,211} For TiO_2 and other metal oxides, metal hydroxides and carbonates can suppress the readsorption of O_2 as superperoxide or peroxide.^{29,40,41,93–96,137}

9. The *crystal structure* of a semiconductor can have a marked influence on the catalytic activity. Active catalysts are generally found to have efficient charge-transport pathways that connect the interior with the surface where water splitting occurs.^{123,156,157,189} Oxide-bridged metal ions that can assume such a role are present in the majority of structures discussed in this review. The structures of LaNbO_4 , BiVO_4 (both Fergusonite), CaWO_4 (Scheelite), and $\text{HT Bi}_2\text{MoO}_6$ contain isolated metal oxide polyhedra, and as a consequence, their catalytic activities are low. Other authors have correlated distortions in metal oxide polyhedra with the catalytic activity. In the case of RuO_2 -modified BaTi_4O_9 , for example, distorted TiO_6 octahedra are believed to cause dipole moments that aid electron-hole separation.⁴² Similar arguments were used for the series $\text{M}_2\text{Ti}_6\text{O}_{13}$ ($\text{M} = \text{Na}, \text{K}, \text{Rb}$)^{42,135} and for $\text{M}_2\text{Sb}_2\text{O}_7$ ($\text{M} = \text{Ca}, \text{Sr}$, Weberite), CaSb_2O_6 , and NaSbO_3 (Ilmenite).^{200,201} For KTaO_3 , the cation size of the Ti, Zr, and Hf dopants could be correlated with the distortion of the structure and the O_2 production rate and inversely with the H_2 production rate.¹⁴⁵ Cavities in the structures of the so-called tunnel titanates $\text{M}_2\text{Ti}_6\text{O}_{13}$ ($\text{M} = \text{Na}, \text{K}, \text{Rb}$)^{42,135} have also been implicated in enhanced electrical contact to the RuO_2 cocatalyst. Because of their small size (~ 1 nm), the cocatalyst particles are believed to fit well into the opening of the tunnels. Finally, the high activities of some of the layered Perovskites are attributed to the incorporation of water into the interlayer space.^{43,169} Water incorporation increases the interfacial area and reduces the necessary distance for charge transfer.

10. Effects of *crystal morphology* and *specific surface area*. Because of their physical relationship, these two variables cannot be separately discussed. In general, an increase in the specific surface area (i.e., a reduction in the crystal size) leads to higher catalytic activities. For example, see $\text{La}_2\text{Ti}_2\text{O}_7$,¹²⁵ $\text{Sr}_3\text{Ti}_2\text{O}_7$,¹²⁶ $\text{K}_2\text{La}_2\text{Ti}_3\text{O}_{10}$,¹³⁴ NaTaO_3 ,¹⁴³ $\text{KCa}_2\text{Nb}_3\text{O}_{10}$,¹⁷² NiO-La:NaTaO_3 ,¹² and $\text{Ba}_5\text{Nb}_4\text{O}_{15}$.¹⁸¹ The same applies to the layered Perovskites, which have the ability to increase their interfacial area by incorporating water into the interlayer space (see point. 9 above).¹⁶⁹ Finally, there are also cases where the surface area does not matter (e.g., ZnS)¹³ or where the activity diminishes with an increase in the surface area (CdS and ZnS).^{13,217} The reasons for the latter behavior are not clear.

In conclusion, over 130 semiconductors are known to catalyze the photochemical water-splitting reaction according to eq 1 or either water oxidation or reduction in the presence of sacrificial agents. Even though the principle activity-controlling factors in semiconductor-heterostructures have been identified, many aspects of the function of inorganic photocatalysts are still unclear. Most importantly, the molecular mechanism of water reduction and oxidation on the semiconductor surface has not yet been elucidated in sufficient detail.^{78,98,100,102,103,106} Many questions about charge transfer between semiconductor and cocatalysts, and its dependence on the structural and electronic features of the interface are still open.^{253–257} The effect of variable material preparations and surface impurities on the catalytic activity of semiconductors (e.g. sulfur and oxide on CdS^{70,71,220}) have not been fully considered. These areas represent significant opportunities for improving water splitting photocatalysts. The development of better catalysts is also going to benefit from recent progress in nanoscience. Quantum size effects can now be used for tailoring both electronic structure and reactivity of nanostructures,^{99,258} and synthetic methods can be employed for controlling the morphology of catalysts down to the nanoscale. In combination with modern analytical techniques (e.g. scanning transmission electron microscopy) for materials characterization, these advances will help to further raise the efficiency of photochemical water splitting catalysts.

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